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Highly Tunable Magnetic Phases in Transition-Metal Dichalcogenide  $\text{Fe}_{1/3+\delta}\text{NbS}_2$

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## Popular Summary

Antiferromagnets are a special class of magnetic materials in which the atomic magnetic moments alternate in direction, resulting in a macroscopically zero net moment. Their consequent insensitivity to external magnetic fields makes these materials promising for building spintronic devices, which store and transfer information in electron spins rather than just charge. One such material, the “van der Waals” antiferromagnet  $\text{Fe}_{1/3+\delta}\text{NbS}_2$ , has shown intriguing spintronic behavior: a

rapid current-induced resistance switching that couples to the antiferromagnetism. But the magnetic properties underlying the resistance were unknown. Here, we report on highly tunable magnetic states in this material that are controlled by magnetic defects. Such defects typically inhibit magnetism because of the disorder, rather than creating new functionality. We perform comprehensive neutron diffraction measurements on crystals with varying abundances of iron. We find that the microscopic magnetic moments arrange themselves into two dramatically distinct configurations at individual atomic sites between the extremes of the iron abundances. These results demonstrate that there are nearly degenerate energies for the two magnetic states involved in the switching. These experimental results are strongly supported by our theoretical calculations. We are thus able to determine that the rapid change of the states as a function of iron ratio underlies the tunable resistance switching, thereby providing crucial insights that form the basis for understanding the spintronic behavior. Our work will stimulate more exploration on such unusual antiferromagnetic spintronic behavior and provide a new avenue for defect-induced controllability of the magnetic state in other van der Waals systems. That should, in turn, enable device scientists to design the very best materials for the spintronic devices of the future.

# Highly Tunable Magnetic Phases in Transition-Metal Dichalcogenide $\text{Fe}_{1/3+\delta}\text{NbS}_2$

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Eran Maniv<sup>1,2,5</sup>, Yiming Qiu<sup>3</sup>, A. A. Aczel<sup>6</sup>, Nicholas S. Settinieri<sup>7</sup>, Jeffrey B. Neaton<sup>1,8,2</sup>,  
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Layered transition-metal dichalcogenides (TMDCs) host a plethora of interesting physical phenomena ranging from charge order to superconductivity. By introducing magnetic ions into  $2H\text{-}TA_2$  ( $T = \text{Nb, Ta}$ ;  $A = \text{S, Se}$ ), the material forms a family of magnetic intercalated TMDCs  $M_xTA_2$  ( $M = 3d$  transition metal). Recently,  $\text{Fe}_{1/3+\delta}\text{NbS}_2$  has been found to possess intriguing resistance switching and magnetic memory effects coupled to the Néel temperature of  $T_N \sim 45$  K [Maniv *et al.*, *Nat. Phys.* **17**, 525 (2021); *Sci. Adv.* **7**, eabd8452 (2021)]. We present comprehensive single crystal neutron diffraction measurements on underintercalated ( $\delta \sim -0.01$ ), stoichiometric, and overintercalated ( $\delta \sim 0.01$ ) samples. Magnetic defects are usually considered to suppress magnetic correlations and, concomitantly, transition temperatures. Instead, we observe highly tunable magnetic long-ranged states as the Fe concentration is varied from underintercalated to overintercalated, that is, from Fe vacancies to Fe interstitials. The under- and overintercalated samples reveal distinct antiferromagnetic stripe and zigzag orders, associated with wave vectors  $k_1 = (0.5, 0, 0)$  and  $k_2 = (0.25, 0.5, 0)$ , respectively. The stoichiometric sample shows two successive magnetic phase transitions for these two wave vectors with an unusual rise-and-fall feature in the intensities connected to  $k_1$ . We ascribe this sensitive tunability to the competing next-nearest neighbor exchange interactions and the oscillatory nature of the Ruderman-Kittel-Kasuya-Yosida mechanism. We discuss experimental observations that relate to the observed intriguing switching resistance behaviors. Our discovery of a magnetic defect tuning of the magnetic structure in bulk crystals  $\text{Fe}_{1/3+\delta}\text{NbS}_2$  provides a possible new avenue to implement controllable antiferromagnetic spintronic devices.

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Subject Areas: Condensed Matter Physics, Magnetism

## I. INTRODUCTION

Layered magnetic van der Waals (vdW) materials have recently attracted tremendous interest, resulting in rapid

progress in fundamental studies of novel vdW physical phenomena together with promising potential for spintronic applications [1–4]. The weak van der Waals bonds make single crystals readily cleavable thereby offering a new platform to study the evolution of the behavior from three dimensions (3D) down to the 2D limit. Moreover, the wide flexibility of 2D atomic samples allows for an efficient manipulation of magnetic states through external perturbations, such as strain, gating, proximity effect, and pressure [5–8]. In bulk magnetic vdW crystals, usually high hydrostatic pressure [9,10] or significant chemical substitution [11,12] is utilized to modulate the magnetic state or the effective dimensionality via tuning of the interlayer exchange couplings. Magnetic defects are typically

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considered to be responsible for inhibiting long-range magnetism due to the atomic-scale disorder. Here we demonstrate novel behavior in which magnetic defects tune the magnetic ground states in the transition-metal dichalcogenide (TMDC) bulk crystal  $\text{Fe}_{1/3+\delta}\text{NbS}_2$ .

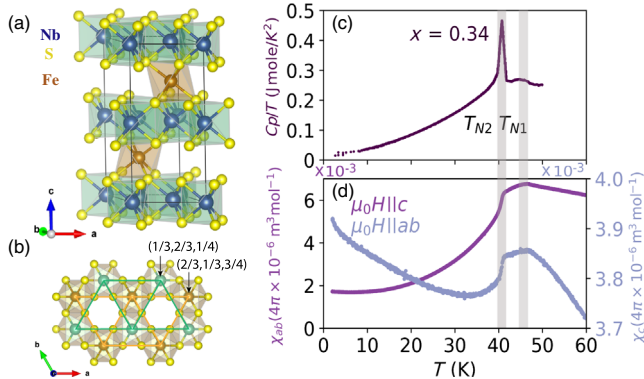
$\text{Fe}_{1/3+\delta}\text{NbS}_2$  is a member of a large class of intercalated TMDCs,  $M_x\text{TA}_2$  family ( $M = 3d$  transition metal;  $T = \text{Nb}, \text{Ta}$ ;  $A = \text{S}, \text{Se}$ ) [13,14]. The host material is a prototypical example of a charge density wave system; recently, these systems have been attracting major attention because of other exotic properties, such as possible quantum spin liquid phases and 2D superconductivity [15–20]. The vdW bonding between chalcogen atoms of adjacent  $2H\text{-TA}_2$  layers allows the ready intercalation of magnetic atoms. The intercalated atoms order into a stacked  $\sqrt{3} \times \sqrt{3}$  superlattice when  $x = 1/3$  [21]. This family of compounds shares the same crystal structure with a noncentrosymmetric space group  $P6_322$  and a bilayer triangular arrangement of the intercalated atoms [Figs. 1(a) and 1(b)]. The broken inversion symmetry results in a Dzyaloshinskii-Moriya (DM) interaction between planes allowing an in-plane moment in addition to the competing bilinear exchange interactions with their concomitant geometric frustration. In addition, as a metallic system, there is a strong interaction between the conduction electrons and the local moments via the Ruderman-Kittel-Kasuya-Yosida (RKKY) mechanism. Depending on the host  $2H\text{-TA}_2$  layer and the intercalated species, the family exhibits a fascinating variety of magnetic and electronic properties [22–30] in bulk samples.

In the intercalated variant  $M_{1/3}\text{NbS}_2$  subgroup, chiral helimagnetism was observed for the Cr and Mn species

[24–28,31]; the V version exhibits a spin structure characterized by ferromagnetic planes stacked antiferromagnetically with canted in-plane moments [30,32]; the Co version shows a stripe order with spins directing in the  $ab$  plane [29]. Novel physical properties were reported in these species, including the anomalous Hall effect, an electrical magnetochiral effect, and magnetic soliton confinement [33–35]. Most materials in this family are characterized by an easy-plane anisotropy and mostly dominant ferromagnetic interactions. In contrast, the intercalated Fe version displays predominantly antiferromagnetic (AFM) correlations and a strong easy-axis anisotropy [13,36,37].

Recently, a resurgence of interest in the Fe version has been sparked by the demonstration of intriguing spintronic properties in bulk  $\text{Fe}_{1/3+\delta}\text{NbS}_2$  crystals [38,39]. Both current-induced resistance switching and magnetic memory effects were reported below the Néel transition temperature  $T_N \sim 45$  K. Moreover, the relevant spintronic properties were found to depend sensitively on the intercalation ratio  $x (=1/3 + \delta)$  [40]. By decreasing the ratio slightly below  $1/3$ , the system exhibited a much more prominent spintronic response concomitant with dramatic spin-glass-like behavior below the AFM Néel temperature. There are so far only a few known examples of current-induced switching behavior in AFM single crystal compounds [41,42]. The mechanism is believed to entail an applied current inducing a spin polarization due to the combination of the breaking of inversion symmetry and Rashba spin-orbit coupling [43]. It has been argued that the reported resistance switching in the off-stoichiometric sample of  $\text{Fe}_{1/3+\delta}\text{NbS}_2$  somehow relates to the observed spin-glass behavior [38,40], thence providing a possible new way to explore AFM spintronic devices. Therefore, a complete understanding of the magnetic ground states and magnetic correlations as a function of the intercalation ratio is essential to uncover the mechanism of the observed interesting spintronic properties. Further, the only relevant information about the magnetic structures which currently exists derives from neutron powder diffraction measurements carried out decades ago at low temperatures [44]. In addition to the spintronic motivation, this system is of intrinsic interest as a vdW material with interesting and, as we shall see, novel magnetic properties.

In this paper, we report detailed neutron scattering measurements on high-quality single crystals of Fe-intercalated TMDC  $\text{Fe}_{1/3+\delta}\text{NbS}_2$  with  $x$  spanning  $1/3$ . Surprisingly, we found highly tunable magnetic phases in the bulk crystal that are more versatile than the single phase reported in previous work. By a comprehensive experimental investigation together with modeling of the magnetic structures, we determined two long-ranged antiferromagnetically ordered states and that one can tune from one state to the other by varying  $x$  subtly from less than to greater than  $1/3$ , that is, by varying from Fe vacancies to Fe interstitials ( $\delta \sim \pm 0.01$ ). The stoichiometric sample with



F1:1 FIG. 1. (a) Crystallographic structure of  $\text{Fe}_{1/3}\text{NbS}_2$ . Fe atoms  
 F1:2 occupy  $2c$  Wyckoff positions, forming a bilayer triangular lattice  
 F1:3 with a noncentrosymmetric space group  $P6_322$ . (b) The view in  
 F1:4 the  $ab$  plane showing only Fe and S atoms. Orange and green  
 F1:5 spheres represent two Fe triangular lattice layers with different  
 F1:6  $c$ -axis coordinates marked in the figure. (c) Representative  
 F1:7 specific heat and (d) magnetization measurement with applied  
 F1:8 field  $\mu_0 H = 0.1$  T ( $T$ ) in the low temperature region for  $x = 0.34$   
 F1:9 sample. The shaded regions mark two anomalies, identified as  
 F1:10 antiferromagnetic transitions by neutron scattering measure-  
 F1:11 ments. For other samples, see the Appendix.

138  $x = 1/3$ , on the other hand, exhibits both magnetic  
 139 structures characterized by two successive magnetic phase  
 140 transitions upon cooling. In Sec. IV, we discuss this  
 141 tunability and its implications to the fascinating spintronic  
 142 behavior exhibited by these materials. This finding is the  
 143 first example of such unusual switching and exchange bias  
 144 behaviors in the intercalated TMDCs  $M_xTA_2$  family; it can  
 145 provide an archetypal case for magnetic defect-induced  
 146 switching of the magnetic state in bulk magnetic vdW  
 147 systems.

## 148 II. METHODS

149 High-quality single crystals were synthesized using a  
 150 chemical vapor transport method with a polycrystalline  
 151 precursor made from Fe, Nb, and S elements in the ratio of  
 152  $x:1:2$  [21]. The crystals used in this paper are not the same  
 153 ones made into devices in Ref. [40], but sizable pieces  
 154 either from the same batch or grown for the neutron  
 155 diffraction experiments by the same recipes and experi-  
 156 mental setups. We weighed the initial Fe powders with  
 157 calculating the ratio of  $x = 0.29, 0.3, 0.32, 0.34$ . The values  
 158 of the resulting intercalation ratio  $x$  of individual single  
 159 crystals were determined by energy dispersive x-ray spec-  
 160 troscopy (EDX) (Fig. 14). By measuring  $\sim 20$  points for  
 161 each crystal, the mean value for each crystal is  $x = 0.31,$   
 162  $0.32, 0.33, 0.35$ . The slightly larger values of the mean ratio  
 163 is due to systematic shifts. The standard deviation for each  
 164 crystal is  $0.003, 0.003, 0.005, 0.003$ , respectively. But  
 165 considering other errors from the instrument, the estimated  
 166 error is  $\pm 0.01$  for our crystals used in neutron scattering.  
 167 The ratio of  $x = 0.34$  crystal was formally determined by  
 168 Maniv *et al.* [38,40]. The actual values of  $x$  were confirmed  
 169 with higher accuracy (to the third decimal place) from  
 170 inductively coupled plasma optical emission spectroscopy  
 171 measurements [38]. Though the change of ratio is subtle,  
 172 the system shows a rapid change in the bulk character-  
 173 izations and spintronic properties [40]. As we see later in  
 174 the paper, the neutron scattering measurements represent a  
 175 surprising tunability of spin structures on the individual  
 176 samples, on the other hand confirming the accuracy of the  
 177 relative values of the ratios up to  $\pm 0.01$ .

178 Room temperature single crystal x-ray diffraction (XRD)  
 179 patterns were measured at the ChemXray facility, UC  
 180 Berkeley. Magnetization measurements were performed  
 181 using a Quantum Design MPMS-3 system. The heat  
 182 capacity was measured in a Quantum Design PPMS system  
 183 [45]. Neutron scattering experiments were carried out at  
 184 several instrumental stations. Single crystal diffraction  
 185 mapping at temperatures  $T = 38$  and  $5$  K with coaligned  
 186 crystals (mosaicity  $\sim 5^\circ$ ) in the range of  $x = 0.32$ – $0.34$   
 187 employed the MACS spectrometer at NCNR [46]. The data  
 188 were collected with  $E_f = 5$  meV with a double focusing  
 189 monochromator and a Be filter placed before and after the  
 190 sample. To investigate accurately the tunable magnetic  
 191 state, single crystal neutron diffraction measurements with

one crystal were carried out on SPINS, BT-7 at NCNR, and  
 192 HB1A at HFIR for different intercalation ratios:  $x = 0.31,$   
 193  $0.32, 0.33, 0.34,$  and  $0.35$  with masses of  $12, 23, 9, 3,$   
 194 and  $27$  mg, respectively. Measurements were conducted with a  
 195 PG (002) monochromator and analyzer using  $E_f = 5, 14.7,$   
 196 and  $14.48$  meV neutrons on SPINS, BT-7, and HB1A,  
 197 respectively. We discuss the density functional theory  
 198 (DFT) calculation strongly related to our experimental  
 199 results. The DFT calculations utilized the Perdew-Burke-  
 200 Ernzerhof (PBE) functional and added a Hubbard  $U$   
 201 correction accounting for the Fe  $d$  electrons. For details  
 202 of the DFT calculations, see Ref. [47].  
 203

## 204 III. EXPERIMENTAL RESULTS

### 205 A. Crystal structures and magnetization

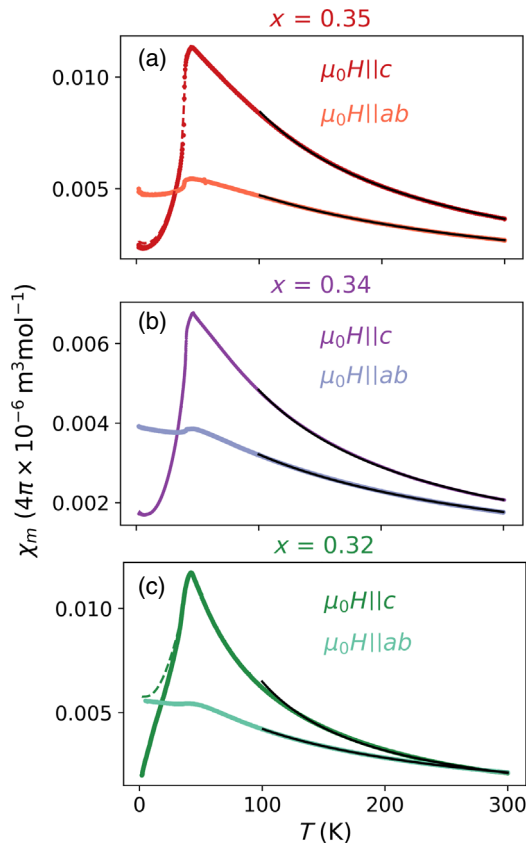
206 The crystallographic structure of  $Fe_{1/3}NbS_2$  is identical  
 207 to that of other species in this family, described by the space  
 208 group  $P6_322$ , with a triangular sublattice of iron ions  
 209 intercalated in the honeycomb  $2H$ - $NbS_2$  [Fig. 1(a)]. One  
 210 crystallographic unit cell contains two equivalent iron sites  
 211 at coordinates  $(1/3, 2/3, 1/4)$  and  $(2/3, 1/3, 3/4)$ , respec-  
 212 tively, associated with the  $2c$  Wyckoff positions. They  
 213 occupy the vacant octahedral sites stacking between the  
 214 prismatic  $NbS_2$  layers and form two triangular superlattice  
 215 planes [Fig. 1(b)]. The shifted stacking between the two  
 216 layers leads to a noncentrosymmetric structure. The single  
 217 crystal x-ray diffraction pattern has been refined in the  
 218 space group  $P6_322$  with  $R_1$  value of  $3.44\%$ , with the atomic  
 219 coordinates listed in Table IV. This is consistent with a  
 220 previous report [21]. As for other intercalated species, the  
 221 intercalated ions are allowed to occupy  $2b$  and  $2d$  sites,  
 222 leading to occupational disorder [48]. However, in our  
 223 Fe materials this issue is not a severe problem. The major  
 224 Fe occupancy occurs at the  $2c$  sites with a minor ratio at the  
 225  $2b$  sites (Table IV in the Appendix). The x-ray diffraction  
 226 pattern manifests sharp three-dimensional peaks in both the  
 227 ( $HOL$ ) and ( $HK0$ ) planes (Fig. 13) and also from neutron  
 228 experiments for all the samples, suggesting a minimum  
 229 Fe lattice disorder.

230 In the slightly off-stoichiometric samples, the noncen-  
 231 trosymmetric structure with the space group  $P6_322$  is  
 232 unchanged from our single crystal XRD analysis. The  
 233 underintercalated sample  $x = 0.32$  reveals the vacancies at  
 234 the  $2c$  Wyckoff positions with a little occupancy at the  $2b$   
 235 sites. The overintercalated sample ( $x = 0.35$ ) allows a  
 236 larger number of occupancy at the  $2b$  sites for additional  
 237 Fe intercalations. Both ratios, where we detected magnetic  
 238 ordering, preserves a majority of well-ordered Fe lattice, also  
 239 demonstrated by the sharp 3D Bragg peaks from neutron  
 240 scattering and the transmission electron spectroscopy mea-  
 241 surements [38]. The  $x = 0.31$  sample shows some disorder  
 242 with the occupancy value of  $0.1$  at the  $2b$  sites (Table IV).  
 243 The detailed information of structure analysis for all the  
 244 ratios are presented in the Appendix. In all, our crystals

245 reveal a homogeneous distribution of Fe atoms (EDX  
 246 measurements in the Appendix) and well-ordered lattice  
 247 with respect to the sharp 3D peaks with the structure  
 248 unaffected by varying the Fe ratio. We also point out that,  
 249 as shown later, the sharp change occurring at particular the  
 250 critical ratio of 1/3 suggests a minimal occupational dis-  
 251 order. At the same time, surprisingly they show rapid  
 252 changes in the magnetization and spin structures determined  
 253 by neutron scattering measurements as described next.

254 The unintercalated host is a *d*-band metal with one  
 255 electron on the Nb ion. Charge transfer from the Fe ions to  
 256 the Nb band results in divalent oxidation states of the  
 257 Fe with localized *d* electrons on the intercalated Fe ions  
 258 [13]. We present magnetic susceptibility and specific  
 259 heat measurements for the  $x = 0.34$  sample in Figs. 1(c)  
 260 and 1(d). Two successive anomalies occur at  $T_{N1} \sim 45$  K  
 261 and  $T_{N2} \sim 41$  K; these features are also observed in the  
 262 specific heat data. Curie-Weiss fits to the magnetic sus-  
 263 ceptibility in the paramagnetic region [Fig. 2(b)] yield  
 264 values for the paramagnetic effective moment  $\mu_{\text{eff}} =$   
 265  $4.3(2)\mu_B$  and Curie-Weiss temperature  $\theta_{\text{CW}} = -49(1)$  K

266 along the *c* axis;  $\mu_{\text{eff}} = 4.0(2)\mu_B$  and  $\theta_{\text{CW}} = -143(2)$  K in  
 267 the *ab* plane. These values are consistent within the range  
 268 of previous reports [22,23,36,44,49] with effective spin  
 269  $S = 2$ . The negative Curie-Weiss temperature suggests that  
 270 antiferromagnetic exchange interactions are dominant. The  
 271 derived single-ion anisotropy *D* is approximately 2 meV  
 272 [50]. In the off-stoichiometric sample with  $x < 1/3$ , one  
 273 transition was identified [40]; and a bifurcation between  
 274 zero-field-cooled (ZFC) and field-cooled (FC) susceptibil-  
 275 ity data was observed, indicating spin-glass-like behavior.  
 276 In the  $x > 1/3$  sample, a small bifurcation between ZFC  
 277 and FC data was observed below  $T_f \sim 10$  K [40]. The  
 278 characterizations of other single crystals used for neutron  
 279 diffraction experiments in this paper are shown in the  
 280 Appendix (Fig. 15). Highly anisotropic magnetization was  
 281 observed in all magnetically ordered samples (Fig. 2). The  
 282 sensitivity to the intercalation ratio *x* of the bulk magnetic  
 283 and thermodynamic properties, as well as the associated  
 284 intriguing spintronic properties, clearly calls for a detailed  
 285 experimental study of the *x* dependence of the magnetic  
 286 ground states in this bilayer triangular lattice system.



F2:1 FIG. 2. Representative magnetization measurements in the full  
 F2:2 temperature region with an applied field  $\mu_0 H = 0.1$  T along the *c*  
 F2:3 axis and in the *ab* plane for (a)  $x > 1/3$ , (b)  $x$  close to 1/3, and  
 F2:4 (c)  $x < 1/3$  sample. The dashed and solid lines correspond to the  
 F2:5 measurements with field-cooled and zero-field-cooled processes,  
 F2:6 respectively. The solid black lines are the results of the Curie-  
 F2:7 Weiss fits with the fitting range between 100 and 300 K.

## B. Neutron scattering measurements

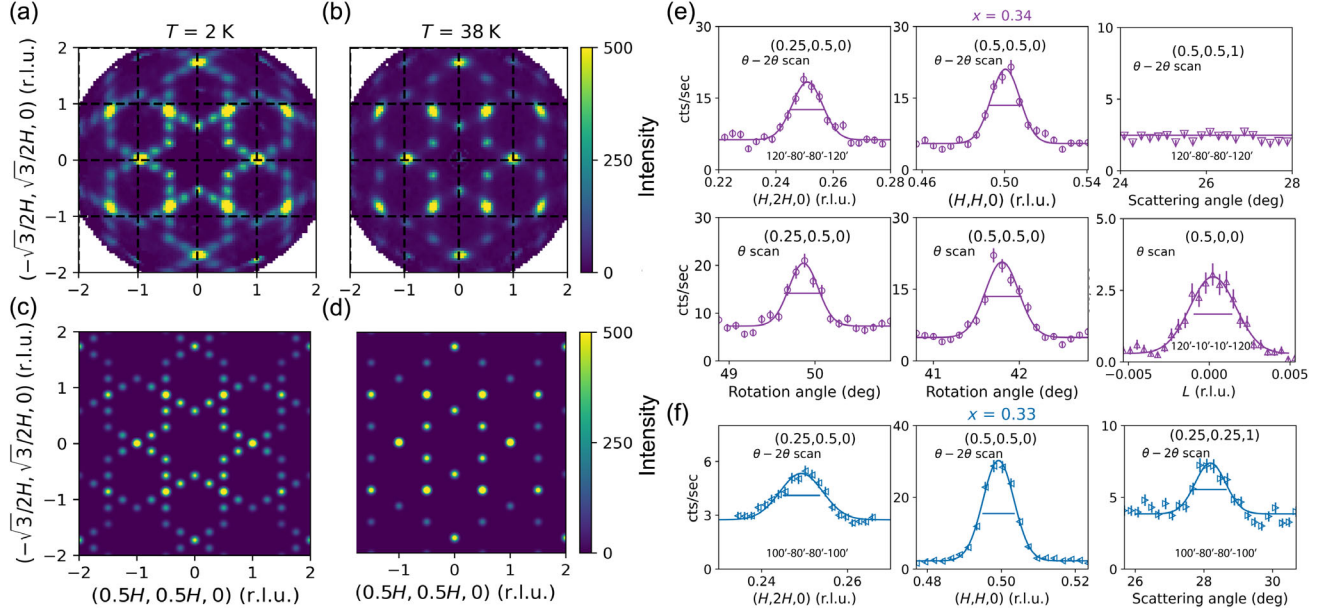
288 We first employed neutron diffraction scattering mea-  
 289 surements in the (*HK*0) scattering plane to study magnetic  
 290 transitions for an assembly of coaligned crystals with *x* in  
 291 the range of 0.32 to 0.34. These experiments were carried  
 292 out at MACS, which is well suited for a broad momentum  
 293 survey. The diffraction pattern, with the data at  $T = 60$  K  
 294 ( $> T_N$ ) subtracted clearly shows two antiferromagnetic  
 295 phases [Figs. 3(a) and 3(b)]. At  $T = 38$  K, superlattice  
 296 peaks are observed at wave vector transfer *Q*'s associated  
 297 with the propagation wave vector  $\mathbf{k}_1 = (0.5, 0, 0)$ . At  
 298  $T = 2$  K, another phase associated with the second propa-  
 299 gation wave vector  $\mathbf{k}_2 = (0.5, 0.25, 0)$  appears leading to  
 300 additional magnetic Bragg peaks. The pattern displays a  
 301 sixfold symmetry; this is the result of three magnetic  
 302 domains with  $Z_3$  symmetry. From the measurements, the  
 303 most intense peaks associated with  $\mathbf{k}_1$  and  $\mathbf{k}_2$  have wave  
 304 vector transfers of  $\mathbf{Q}_1 = (0.5, 0.5, 0)$  (or 6 equivalent  
 305 positions) and  $\mathbf{Q}_2 = (0.25, 0.5, 0)$  (or 12 equivalent posi-  
 306 tions), respectively.

307 The measurement on MACS were carried out on a set  
 308 of coaligned single crystals. To obtain more information  
 309 and specifically to elucidate the *x* dependence of the  
 310 magnetic structures, we measured individual high-quality  
 311 single crystals close to stoichiometry ( $x = 0.33, 0.34$ ),  
 312 underintercalated ( $x = 0.31, 0.32$ ), and overintercalated  
 313 ( $x = 0.35$ ).

### 1. Nearly stoichiometric $x = 1/3$ sample

314 We measured two crystals with  $x = 0.33$  and 0.34  
 315 separately at SPINS and BT7. Representative transverse  
 316 and longitudinal, namely,  $\theta$  and  $\theta - 2\theta$ , scans at 5 K are  
 317 shown in Figs. 3(e) and 3(f). For the  $x = 0.34$  sample, the  
 318





F3:1 FIG. 3. (a),(b) Symmetrized single crystal neutron diffraction patterns collected at MACS by coaligned crystals of  $\text{Fe}_x\text{NbS}_2$   
 F3:2 ( $x = 0.32\text{--}0.34$ ) mounted in the  $(HK0)$  scattering plane at  $T = 2$  and  $38$  K. A dataset acquired at  $T = 60$  K was subtracted as the  
 F3:3 background. The intensity is in the unit of counts per  $3 \times 10^4$  monitor counts. (c),(d) Calculated diffraction patterns for given spin  
 F3:4 configurations. The details are described in the main text. Representative transverse ( $\theta$ ) and longitudinal ( $\theta - 2\theta$ ) scans (dots) measured  
 F3:5 on one single crystal close to stoichiometric ratio with (e)  $x = 0.34$  at BT7 and (f)  $x = 0.33$  at SPINS with  $T = 5$  K. The markers  $\circ$ ,  $\nabla$ ,  
 F3:6  $\triangle$  denote the data collected in the  $(HK0)$ ,  $(HHL)$ ,  $(HOL)$  scattering planes at BT7, and  $\triangleleft$ ,  $\triangleright$  are used for the  $(HK0)$  and  $(HHL)$   
 F3:7 scattering planes at SPINS, compatible with the markers in Fig. 10. The horizontal bars denote the instrument  $Q$  resolutions. The solid  
 F3:8 lines are the results of the fits to a Gaussian line shape. The corresponding configurations of the collimations are written in the panel.  
 F3:9 Error bars in all figures represent one standard deviation.

319 magnetic peak at  $\mathbf{Q}_2 = (0.25, 0.5, 0)$  has comparable  
 320 intensity with the peak at  $\mathbf{Q}_1 = (0.5, 0.5, 0)$  [Fig. 3(e)].  
 321 Both peaks have their full width at half maximum (FWHM)  
 322 determined by the instrumental  $Q$  resolution, marked by  
 323 horizontal bars in the plots. The  $\theta$  scan with the crystal  
 324 in the  $(HOL)$  plane, equivalent to an  $L$  scan, displays also  
 325 a resolution-limited peak [Fig. 3(e)], indicating three-  
 326 dimensional long-range order even though the structures  
 327 are lamellar. For the  $x = 0.33$  sample the relative intensity  
 328 of peaks between  $\mathbf{Q}_1$  and  $\mathbf{Q}_2$  [Fig. 3(f)] are dramatically  
 329 different from that with  $x = 0.34$ , having more intensity  
 330 related to  $k_1 = (0.5, 0, 0)$ . We also collected superlattice  
 331 peaks at a series of reciprocal lattice positions,  $(0.5, 0.5, L)$ ,  
 332 by varying  $L$ . The intensity decreases gradually with  
 333 increasing  $L$  value; the intensity following roughly the  
 334 square of the magnetic form factor, manifesting the  
 335 magnetic nature of the superlattice peaks.

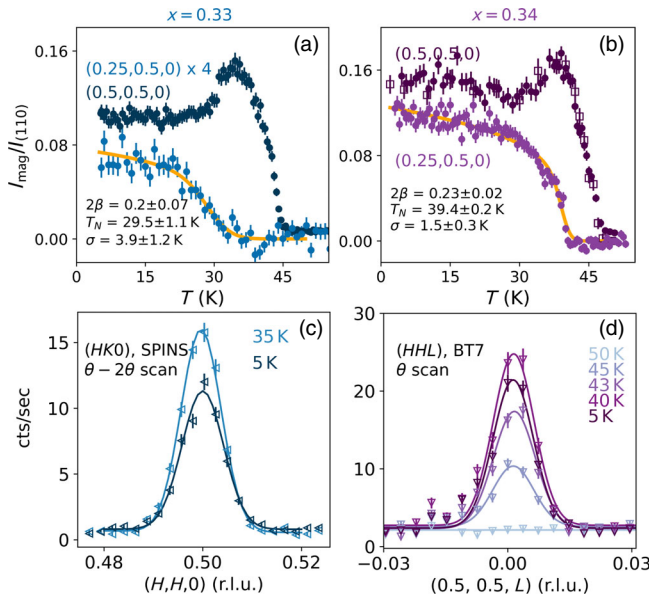
336 To study the temperature evolution of the two magnetic  
 337 phases, the intensities at peak position  $\mathbf{Q}_1$  and  $\mathbf{Q}_2$  were  
 338 measured as a function of temperature for the two samples,  
 339 as shown in Figs. 4(a) and 4(b). The magnetic peak  
 340 intensities are scaled to comply with the integrated areas  
 341 of the peaks measured from the motor scans, and normal-  
 342 ized by the integrated area of the nuclear Bragg peak (110).  
 343 The samples were measured in the  $(HK0)$  scattering plane  
 344 for these two plots. Both samples display the onsets of two

345 magnetic transitions, consistent with the transition temper-  
 346 ature anomalies observed in the bulk characterization  
 347 measurements. The first transition is identified at  $T_{N1} \sim$   
 348  $45$  K based on a guide to the eye. To extract the power law  
 349 exponent  $2\beta$  and  $T_{N2}$ , we assume a Gaussian distribution of  
 350 transition temperatures within the bulk crystal in the power  
 351 law function [51,52]:

$$\int_0^\infty \left(1 - \frac{T}{t_N}\right)^{2\beta} \frac{1}{\sqrt{2\pi}\sigma} e^{-(t_N - T)^2/2\sigma^2} dt_N. \quad (1)$$

352 The fits provide the results  $T_{N2} = 30(1)$  and  $39.4(2)$  K with  
 353 the thermal width of  $\sigma = 4(1)$  and  $1.5(3)$  K, and the power  
 354 law exponent  $2\beta = 0.20(7)$  and  $0.23(2)$  for the  $x = 0.33$   
 355 and  $0.34$  crystals, respectively. The values for  $2\beta$  are close  
 356 to that for the ideal 2D Ising model,  $2\beta = 0.25$ , although,  
 357 because of the large spread in  $T_N$ , one should not over-  
 358 interpret this result. Specifically, we cannot rule out a  
 359 weakly first-order transition.  
 360

361 Interestingly, both nearly stoichiometric samples display  
 362 an increase of the magnetic peak intensity at  $\mathbf{Q}_1$  below  $T_{N1}$ ,  
 363 followed by a partial drop of the intensity below  $T_{N2}$ . This  
 364 rules out the scenario that the stoichiometric sample is  
 365 simply composed of partial under- and overintercalated  
 366 regions; otherwise we should simply see two separated  
 367 order parameter curves. This unusual feature is also



F4:1 FIG. 4. Temperature dependence of the magnetic peak intensities at  $\mathbf{Q}_2 = (0.25, 0.5, 0)$  and  $\mathbf{Q}_1 = (0.5, 0.5, 0)$  for two nearly stoichiometric samples (a)  $x = 0.33$  on SPINS and (b)  $x = 0.34$  on BT7. The peak intensities are scaled to match the integrated intensities (empty squares) and both are normalized by the integrated intensity of the nuclear peak (110). Orange lines are the results of fits to the power law function with a thermal Gaussian distribution of  $T_N$  [ $I \propto \int_0^\infty [1 - (T/T_N)]^{2\beta} \times [1/(\sqrt{2\pi}\sigma)] e^{-(t_N - T_N)^2/2\sigma^2} dt_N$ ] [51,52]. Representative motor scans at  $\mathbf{Q}_1$  at elevated temperatures: (c)  $\theta - 2\theta$  scan in the  $(HK0)$  plane for  $x = 0.33$  and (d)  $\theta$  scan in  $(HHL)$  plane for  $x = 0.34$ . These correspond to in-plane and out-of-plane  $Q$  scans along the  $HH$  and  $L$  directions, respectively. The solid lines are the results of fits to a Gaussian function. Error bars in all panels represent one standard deviation.

368 confirmed in the  $\theta - 2\theta$  and  $\theta$  motor scans at elevated  
 369 temperatures in Figs. 4(c) and 4(d). These measurements  
 370 were carried out in the spectrometer configuration with  
 371 the crystal mounted in the  $(HK0)$  and  $(HHL)$  planes.  
 372 Correspondingly, motor scans traversing across  $\mathbf{Q}_1$  are  
 373 equivalent to scans along the  $HH$  and  $L$  directions,  
 374 respectively. The magnetic peaks at intermediate temper-  
 375 atures ( $T = 35$  K in  $x = 0.33$  and 40 K in  $x = 0.34$ ) show  
 376 higher intensities than the data at 5 K and a constant  
 377 resolution-limited width from the Gaussian peak fits. These  
 378 results preclude explanations due to the change of the  
 379 magnetic correlations from 3D to 2D with decreasing  
 380 temperature, which can lead to the broadening of the peak  
 381 in the out-of-plane direction thereby reducing the peak  
 382 intensity simultaneously within the plane.

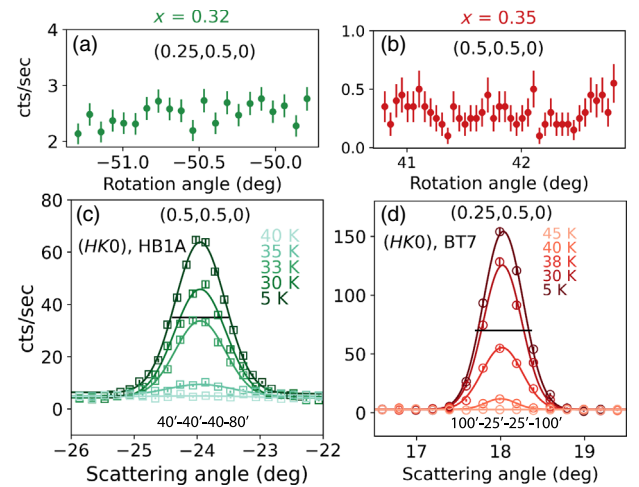
## 2. Off-stoichiometric samples

383  
 384 To investigate the magnetic states and spin-glass-like  
 385 physics in the off-stoichiometric samples, we measured  
 386 two underintercalated samples ( $x = 0.31, 0.32$ ) and one

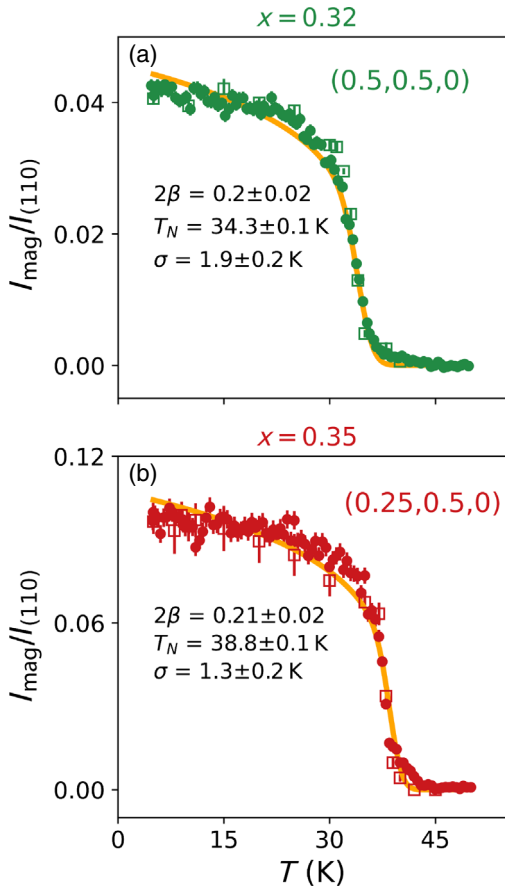
overintercalated sample ( $x = 0.35$ ) in  $\text{Fe}_x\text{NbS}_2$ . The results  
 387 turn out to be quite striking. In the  $x = 0.32$  sample, we  
 388 observed only magnetic peaks associated with wave vector  
 389  $\mathbf{k}_1$ , and no detectable peaks at the positions related to  $\mathbf{k}_2$   
 390 [Figs. 5(a) and 5(c)]. In contrast, we observed only peaks  
 391 associated with  $\mathbf{k}_2$ , not with  $\mathbf{k}_1$ , in the  $x = 0.35$  sample  
 392 [Figs. 5(b) and 5(d)]. The strongest intensity is observed at  
 393  $\mathbf{Q}_1 = (0.5, 0.5, 0)$  and  $\mathbf{Q}_2 = (0.25, 0.5, 0)$ , respectively,  
 394 for each sample. These peak positions were used to study  
 395 the temperature-dependent behavior for each crystal.  
 396

The onset of the peaks at the two positions upon cooling  
 397 clearly manifests magnetic transitions. (Fig. 6). From fits to  
 398 the Gaussian-broadened power law function [Eq. (1)], we  
 399 obtain  $T_N$  of 34.2(1) and 38.8(1) K with widths  $\sigma$  of 1.9(2)  
 400 and 1.3(2) K, and power law exponents  $2\beta$  of 0.20(2) and  
 401 0.21(2) for the  $x = 0.32$  and 0.35 samples, respectively.  
 402 The transition temperature in the  $x = 0.32$  crystal is  
 403 consistent with the second kink of  $\chi_{ab}$  [see Fig. 15(c) in  
 404 the Appendix]. While in  $x = 0.35$ , the transition temper-  
 405 ature coincides with the peak anomaly in the susceptibility  
 406 measurement [40]. The extracted values of the power law  
 407 exponents, as well as for the stoichiometric sample, are  
 408 consistent with the value for the 2D Ising system  
 409 ( $2\beta = 0.25$ ) [53,54] as we noted previously.  
 410

The width of the magnetic Bragg peaks in both samples  
 411 agrees within the measurement uncertainties with the  
 412 instrumental resolution, thence implying long-range  
 413 AFM order. Naively, this might be seen to be unexpected  
 414 since the magnetization measurements manifest a bifurca-  
 415 tion between the ZFC and FC processes and a slow  
 416 relaxation of the magnetization. Specifically, we might  
 417



F5:1 FIG. 5. Sample rotation  $\theta$  scans at the given positions at  
 F5:2  $T = 5$  K for (a)  $x = 0.32$  and (b)  $x = 0.35$ , showing no detect-  
 F5:3 able signals. Representative temperature-dependent  $\theta - 2\theta$  scans  
 F5:4 for (c)  $x = 0.32$  at  $\mathbf{Q}_1 = (0.5, 0.5, 0)$  and (d)  $x = 0.35$  at  
 F5:5  $\mathbf{Q}_2 = (0.25, 0.5, 0)$ . The solid lines are results of fits to the  
 F5:6 Gaussian function with the resolution shown in the horizontal  
 F5:7 black line at 5 K.



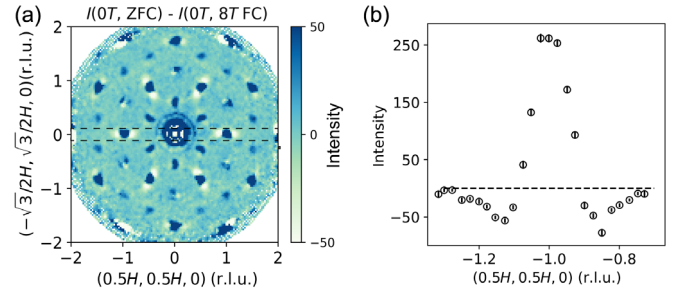
F6:1 FIG. 6. Temperature dependence of the magnetic peak intensity  
 F6:2 (filled dots) for (a) underintercalated sample  $x = 0.32$  at  $\mathbf{Q}_1 =$   
 F6:3  $(0.5, 0.5, 0)$  on HB1A and (b) overintercalated sample  $x = 0.35$   
 F6:4 at  $\mathbf{Q}_2 = (0.25, 0.5, 0)$  on BT7. Empty squares are integrated  
 F6:5 intensities extracted from the Gaussian fits to the  $\theta - 2\theta$  scans in  
 F6:6 Fig. 5. Orange lines are the results of fits to the power law  
 F6:7 function with a Gaussian distribution of  $T_N$  [51].

418 have expected to observe a short-ranged magnetically  
 419 ordered state in light of the apparent spin-glass behavior.

420 In the  $x = 0.31$  crystal, we examined scans along the  
 421 high symmetry directions and also carried out a 2D  
 422 mapping in  $(HK0)$  plane at 5 K on SPINS. To our surprise,  
 423 we found no short- or long-ranged magnetic signal above  
 424 the background level below  $T_f$  or  $T_N$ . This could be due  
 425 to the in-plane disorder that destroys magnetic order, or that  
 426 the magnetic signals were sufficiently broad that they could  
 427 not be distinguished from the background.

### 428 3. Field-cooled neutron scattering measurements

429 We also employed neutron diffraction measurements in  
 430 the presence of an applied magnetic field at MACS on the  
 431 coaligned crystals to investigate any relevant spin-glass  
 432 behavior. Interestingly, we observed a broadening of the  
 433 magnetic superlattice peak at wave vector  $\mathbf{Q}_1$  by cooling  
 434 the crystal across  $T_N$  under an 8 T ( $T$ ) magnetic field. This



F7:1 FIG. 7. (a) Single crystal neutron diffraction pattern in  $(HK0)$   
 F7:2 plane at 2 K with zero-field-cooled process, with data collected  
 F7:3 with cooling in an 8 T ( $T$ ) magnetic field subtracted. The data  
 F7:4 were folded with sixfold rotational symmetry and expanded to the  
 F7:5 full rotation angle for presentation purposes. The intensity is in  
 F7:6 the unit of counts per  $3 \times 10^4$  monitor counts. (b) The cut along  
 F7:7  $(0.5H, 0.5H, 0)$  with integrating the range of  $H = [-0.07, 0.07]$   
 F7:8 r.l.u. in  $(-\sqrt{3}/2H, \sqrt{3}/2H, 0)$  direction, as marked by the dashed  
 F7:9 rectangle in (a). These are the data obtained by subtracting the 8 T  
 F7:10 field-cooled measurement. The dashed line denotes the base line  
 F7:11 with zero intensity in (b).

435 broadening is evident by viewing the diffraction pattern in a  
 436 ZFC measurement with the pattern obtained after sub-  
 437 tracting an equivalent FC measurement as shown in Fig. 7.  
 438 The  $Q$  cut across the position of  $(-0.5, -0.5, 0)$  in the  
 439 difference pattern exhibits more intensity at the peak center  
 440 and two symmetric wings with negative net counts after the  
 441 subtraction [Fig. 7(b)]. That implies a different line shape  
 442 of the magnetic peak in the FC process compared with the  
 443 ZFC process. Such peak broadening on field cooling was  
 444 also observed in other dilute two-dimensional Ising anti-  
 445 ferromagnets [52,55], in which the broadening is attributed  
 446 to the random staggered magnetic field generated by the  
 447 applied magnetic field.

448 To sum up our single crystal neutron scattering mea-  
 449 surements, we have obtained the following principal  
 450 magnetic properties in  $\text{Fe}_x\text{NbS}_2$  with varying intercalation  
 451 ratio  $x$  but with identical crystallographic structures.  
 452 (1) Strong magnetic intensities at in-plane positions sug-  
 453 gesting that the spins are oriented along the  $c$  axis,  
 454 consistent with the highly anisotropic magnetization data.  
 455 (2) Two types of magnetic phases associated with wave  
 456 vector  $\mathbf{k}_1 = (0.5, 0, 0)$  and  $\mathbf{k}_2 = (0.25, 0.5, 0)$  were  
 457 observed. We observed magnetic peaks related to only  
 458  $\mathbf{k}_1$  in samples with  $x < 1/3$ , both  $\mathbf{k}_1$  and  $\mathbf{k}_2$  in stoichi-  
 459 ometric samples,  $x \sim 1/3$ , and  $\mathbf{k}_2$  alone in overintercalated  
 460 crystals, that is,  $x > 1/3$ . (3) In crystals with  $x \sim 1/3$ , there  
 461 are two successive magnetic transitions, showing a rise-  
 462 and-fall feature in the peak intensity curve. (4) All samples,  
 463 except for the heavily underintercalated sample ( $x = 0.31$ ),  
 464 exhibit resolution-limited peaks implying long-range order  
 465 within the given resolution. The fitted power law exponent  
 466  $\beta$  is consistent with 2D Ising behavior ( $\beta = 0.125$ ) [54],  
 467 although the uncertainties are large and we cannot rule out  
 468 a weakly first-order transition due to the spread of the

469 transition temperature  $T_N$ . (5) No clear features related to  
 470 spin-glass physics are evident from neutron diffraction  
 471 measurements under zero field; however, magnetic peak  
 472 broadening in the measurement with coaligned crystals  
 473 presumably due to induced staggered random field effects  
 474 was observed in the field-cooled process. We note that the  
 475 dramatic changes in the magnetic structures as going from  
 476  $x = 0.32$  to  $x = 0.35$  necessitate that the crystals are  
 477 homogeneous with variation of  $x$  of at most 0.01 within  
 478 the samples.

### 479 C. Magnetic structure determination

480 For a systematic analysis of possible magnetic structures  
 481 associated with  $\mathbf{k}_1$  and  $\mathbf{k}_2$ , we use representation analysis  
 482 in SARAH [56] and also BasIreps in FullProf [57], and  
 483 calculate magnetic scattering intensities. Given the crys-  
 484 tallographic symmetry  $P6_322$ , the  $2c$  Wyckoff position for  
 485 the Fe atoms, and propagation wave vector  $\mathbf{k}$ , group theory  
 486 analysis describes that the magnetic representation  $\Gamma_{\text{mag}}$  can  
 487 be decomposed into irreducible representations (IRs) and  
 488 their corresponding basis vectors (BVs). According to  
 489 Landau theory, the magnetic symmetry can be described  
 490 by one IR for each transition. This information is then  
 491 implemented to perform model calculations for the deter-  
 492 mination of the magnetic structure. For single crystal  
 493 diffraction, the measured magnetic coherent cross section  
 494 follows the expression [58]:

$$\frac{d\sigma}{d\Omega} = N_M \frac{(2\pi)^3}{V_M} p^2 \sum_{\mathbf{G}_M} \delta(\mathbf{Q} - \mathbf{G}_M) |\mathbf{F}_\perp(\mathbf{Q})|^2. \quad (2)$$

496 Here  $|\mathbf{F}_\perp(\mathbf{Q})|^2 = |\mathbf{F}_M(\mathbf{Q})|^2 - |\hat{\mathbf{e}} \cdot \mathbf{F}_M(\mathbf{Q})|^2$  contains the  
 497 static magnetic structure factor and magnetic form factor  
 498 and represents the component of the spin axis perpendicular  
 499 to  $\mathbf{Q}$ .  $\mathbf{G}_M$  is the wave vector transfer associated with the  
 500 reciprocal lattice vector  $\boldsymbol{\tau}$  as  $\mathbf{G}_M = \boldsymbol{\tau} \pm \mathbf{k}$  and a single  
 501 propagation vector  $\mathbf{k}$ .  $N_M$  and  $V_M$  are the number and  
 502 volume of the magnetic unit cell, respectively, and  
 503  $p = 2.695$  fm. The magnetic structure factor  $\mathbf{F}_M$  is related  
 504 to the spin configuration as

$$\mathbf{F}_M(\mathbf{Q}) = \sum_j f(\mathbf{Q}) \mathbf{S}_{\mathbf{k},j} e^{i\mathbf{Q} \cdot \mathbf{r}_j}, \quad (3)$$

506 where  $\mathbf{S}_{\mathbf{k},j}$  is the spin moment for atom  $j$  at the position  $r_j$   
 507 within a magnetic unit cell, and can be written by the BVs in  
 508 irreducible representation analysis. By this formalism we can  
 509 calculate magnetic scattering intensities for different spin  
 510 structures and determine the configuration most accordant  
 511 with the data.

512 First, we describe the representation analysis for the two  
 513 types of propagation vectors  $\mathbf{k}_1$  and  $\mathbf{k}_2$  and discuss the  
 514 choice of BVs supported by the observed data.

### 1. Phase $\mathbf{k}_1 = (0.5, 0, 0)$

515

For the propagation vector  $\mathbf{k}_1 = (0.5, 0, 0)$ , the magnetic  
 516 representation  $\Gamma_{\text{mag}}$  can be decomposed into IRs  $\Gamma_{\text{mag}} =$   
 517  $\Gamma_1 + 2\Gamma_2 + 2\Gamma_3 + \Gamma_4$  with corresponding BVs listed in  
 518 Table I. Because the moment direction has been determined  
 519 to be predominantly along the  $c$  axis by both the magnetic  
 520 susceptibility and neutron data, only  $\Gamma_2$  ( $\psi_2, \psi_3$ ) and  $\Gamma_3$   
 521 ( $\psi_4, \psi_5$ ) are relevant. For the same reason, we concentrate  
 522 on the BV  $\psi_3$  and  $\psi_5$ . The difference between them is two  
 523 Fe atoms in one unit cell [Fig. 1(a)] oriented parallel or  
 524 antiparallel, respectively. The calculated magnetic scatter-  
 525 ing patterns [Fig. 3(d) and Fig. 16 in the Appendix] with  $\psi_5$   
 526 agree with the data, showing an antiparallel stacking  
 527 between two Fe spins. This is consistent with the strongest  
 528 intensity being observed at  $\mathbf{Q}_1 = (0.5, 0.5, 0)$ . Though  $\psi_4$   
 529 is also allowed by group theory analysis, however, no  
 530 peak feature is observed at the position  $\mathbf{Q} = (0.5, 0.5, 1)$   
 531 [Fig. 3(e)] disfavoring the spin component related to that  
 532 peak position, suggesting an absence of any measurable in-  
 533 plane moment associated with  $\mathbf{k}_1$ .  
 534

The spin configuration corresponding to  $\psi_5$  is shown  
 535 in Figs. 8(a)–8(c). It consists of spins oriented in the  
 536 same direction along one crystal axis and alternating  
 537 along the other one, forming a stripe pattern elongated  
 538 along an in-plane high symmetric crystal axis. Two Fe  
 539 atoms with different  $c$  coordinates stack antiferromag-  
 540 netically. We named this configuration ‘‘AFM stripe’’ for  
 541 simplicity. The magnetic unit cell is 2 times the size of  
 542 the structural unit cell. Note that there are three equiv-  
 543 alent  $k$  vectors [(0.5,0,0), (0,0.5,0), and (0.5,−0.5,0)],  
 544 corresponding to three magnetic domains along three  
 545 directions [Fig. 8(c)].  
 546

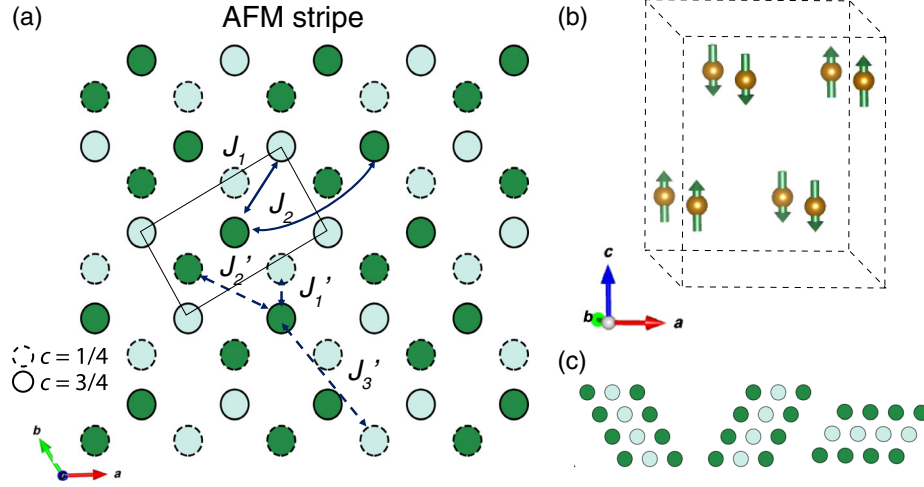
### 2. Phase $\mathbf{k}_2 = (0.25, 0.5, 0)$

547

For the propagation vector  $\mathbf{k}_2 = (0.25, 0.5, 0)$ , the  
 548 magnetic representation  $\Gamma_{\text{mag}}$  decomposes into IRs  
 549  $\Gamma_{\text{mag}} = 3\Gamma_1 + 3\Gamma_2$  with corresponding BVs listed in  
 550 Table II. Six BVs describe a collinear ( $\psi_2, \psi_3, \psi_5, \psi_6$ )  
 551 and noncollinear ( $\psi_1, \psi_4$ ) spin configuration.  $\psi_2$  and  $\psi_6$   
 552 depict two parallel Fe spins, while  $\psi_3$  and  $\psi_5$  represent  
 553

TABLE I. Basis vectors (BVs)  $\psi_i$  of IRs for two Fe atoms in the unit cell [Fe1, (0.333,0.667,0.25); Fe2, (0.667,0.333,0.75)] associated with propagation vector  $\mathbf{k}_1 = (0.5, 0, 0)$ . BVs are defined by the crystallographic axes.

IR	BV	Fe1	Fe2
$\Gamma_1$	$\psi_1$	(2 1 0)	(−2 −1 0)
$\Gamma_2$	$\psi_2$	(0 −1 0)	(0 1 0)
	$\psi_3$	(0 0 1)	(0 0 1)
$\Gamma_3$	$\psi_4$	(0 −1 0)	(0 −1 0)
	$\psi_5$	(0 0 1)	(0 0 −1)
$\Gamma_4$	$\psi_6$	(2 1 0)	(2 1 0)



F8:1 FIG. 8. AFM stripe magnetic structure associated with the  $\mathbf{k}_1 = (0.5, 0, 0)$  domain in  $\text{Fe}_x\text{NbS}_2$  ( $x < 1/3$ ): view in (a)  $ab$  plane and  
 F8:2 (b) three dimensions. AFM in the notation for the spin configuration is defined when two Fe atoms in one unit cell have antiparallel  
 F8:3 spins. Circles with solid and dashed outlines in (a) represent two Fe layers at  $c = 3/4$  and  $c = 1/4$ . Dark and light colors denote spins up  
 F8:4 and down. Solid rectangle depicts the smallest magnetic unit cell. (c) Plots of three equivalent domain directions within one Fe triangular  
 F8:5 lattice layer.

554 antiparallel spins in one unit cell. By qualitatively compar-  
 555 ing these to the diffraction pattern associated with  $\mathbf{k}_2$   
 556 [Fig. 3(a)], the calculated patterns [Fig. 3(c) and Fig. 17 in  
 557 the Appendix] that are described by  $\psi_3$  and  $\psi_5$  clearly  
 558 follow selection rules for the magnetic peaks that are  
 559 consistent with the observation. The other BVs result in  
 560 unwanted reflections, for example,  $\mathbf{Q} = (0.25, 0.25, 0)$ .

561 The corresponding spin configuration is displayed in  
 562 Figs. 9(a)–9(c). The difference between  $\psi_3$  and  $\psi_5$  is spin  
 563 moments directed out of plane and in plane, respectively.  
 564 Within the layer, spins point in the sequence of  $++--$   
 565 along one crystal axis ( $+$  and  $-$  denote spins up and  
 566 down for  $\psi_3$ ). Two Fe atoms in one unit cell have spins  
 567 pointing in opposite directions. Since connecting the same  
 568 direction of the Fe spins within one layer institutes a zigzag  
 569 route, we named this configuration “AFM zigzag” for  
 570 simplicity. The minimum magnetic unit cell is 4 times  
 571 the structural unit cell, and orthohexagonal. Note that there  
 572 are six equivalent  $k$  vectors [(0.5,0.25,0), (0.5,−0.75,0),

(0.75,−0.25,0), (0.25,0.5,0), (−0.75,0.5,0), and  
 (−0.25,0.75,0)], leading to three magnetic domains along  
 three directions [Fig. 9(c)].

Next, we quantitatively determine the magnetic struc-  
 tures for off and nearly stoichiometric  $\text{Fe}_x\text{NbS}_2$  samples.

### 3. Spin structure

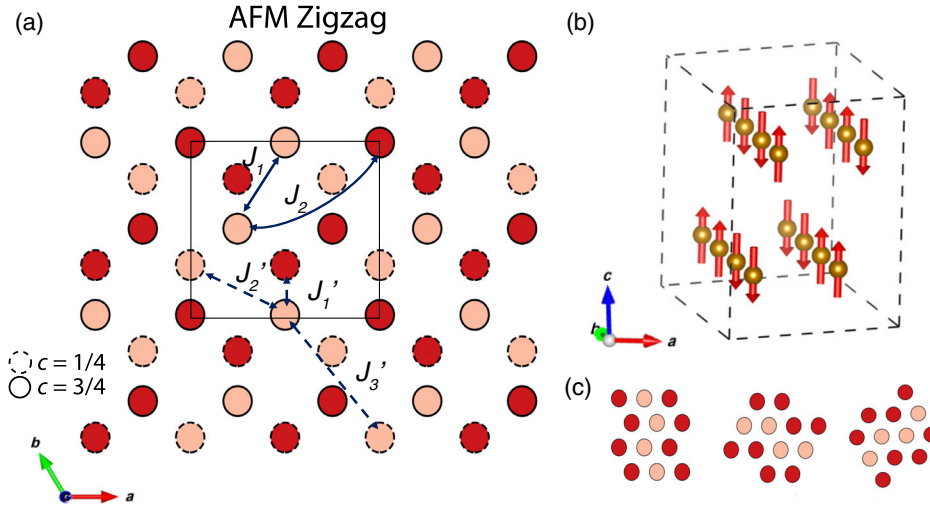
The neutron scattering data for the off-stoichiometric  
 samples display a single magnetic transition with wave  
 vector  $\mathbf{k}_1$  and  $\mathbf{k}_2$  in the  $x = 0.32$  and  $x = 0.35$  crystals,  
 respectively. In the  $x = 0.32$  sample, since the strongest  
 peak in the  $(HK0)$  plane is at  $\mathbf{Q} = (0.5, 0.5, 0)$ , the spin  
 structure with spins along the  $c$  axis is described by the  
 basis vector of  $\psi_5$  in the irreducible representation  $\Gamma_3$   
 (Table I). The ordered moment was obtained as  $m =$   
 $2.6(3)\mu_B$  from comparison between the observed and  
 calculated intensities [Fig. 10(c)] by using Eqs. (2) and (3)  
 and a normalization factor from the nuclear peaks. The spin  
 configuration can be described as AFM stripe with the  
 moments oriented along the  $c$  axis [Fig. 8(b)].

In the overintercalated  $x = 0.35$  sample, all of the  
 magnetic reflections are related to the wave vector  $\mathbf{k}_2$ .  
 Since only one IR is allowed for a second-order phase  
 transition,  $\psi_3$  in  $\Gamma_1$  (Table. II) was assigned to provide  
 consistent results with the observed magnetic intensities  
 [Fig. 10(e)]. The ordered moment was obtained as  
 $m = 3.0(3)\mu_B$ , and the spin configuration can be described  
 as AFM zigzag with moments along  $c$  axis [Fig. 9(b)].

Nearly stoichiometric samples with  $x \sim 1/3$  have two  
 magnetic transitions. Below  $T_{N1}$ , the spin structure can be  
 ascribed to the AFM stripe configuration [Fig. 8(b)]  
 depicted by  $\psi_5$  in  $\Gamma_3$  with ordered moment of  $2.9(3)\mu_B$   
 [Fig. 10(a)]. Below  $T_{N2}$ , to elaborate the rise-and-fall

TABLE II. Basis vectors  $\psi_i$  of IRs for two Fe atoms in unit cell  
 [Fe1, (0.333,0.667,0.25); Fe2, (0.667,0.333,0.75)] associated  
 with propagation vector  $\mathbf{k}_2 = (0.25, 0.5, 0)$ . BVs are defined  
 by the crystallographic axes.

IR	BV	Fe1	Fe2
$\Gamma_1$	$\psi_1$	(1 0 0)	(−i −i 0)
	$\psi_2$	(0 1 0)	(0 i 0)
	$\psi_3$	(0 0 1)	(0 0 −1)
$\Gamma_2$	$\psi_4$	(1 0 0)	(i i 0)
	$\psi_5$	(0 1 0)	(0 −i 0)
	$\psi_6$	(0 0 1)	(0 0 i)

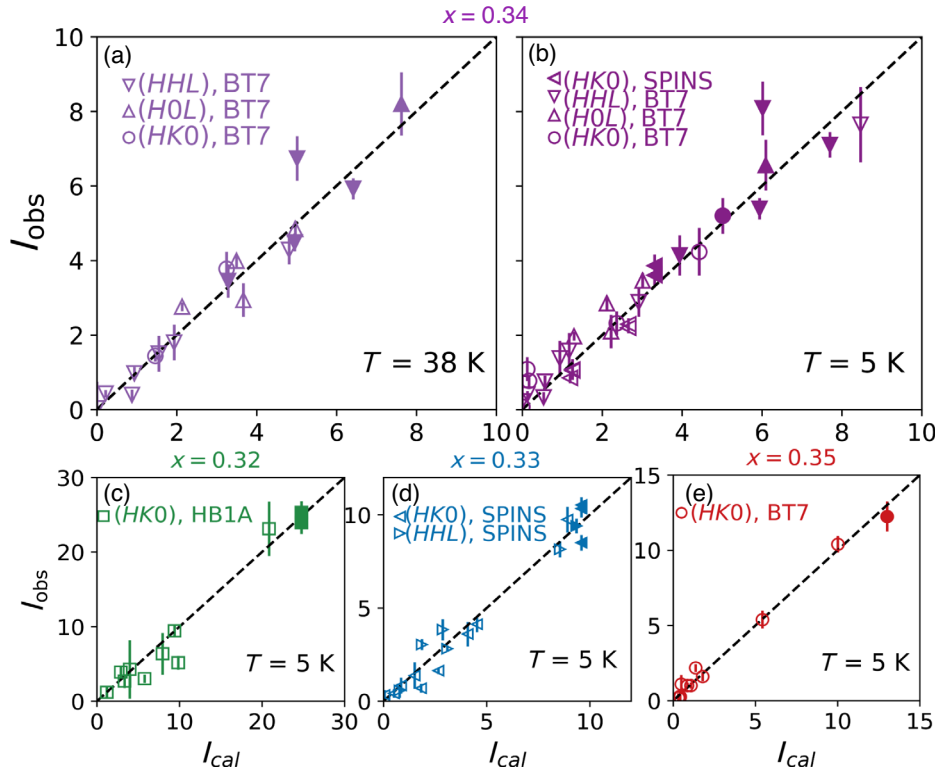


F9:1 FIG. 9. AFM zigzag magnetically ordered phase associated with the  $\mathbf{k}_2 = (0.5, 0.25, 0)$  domain in  $\text{Fe}_x\text{NbS}_2$  ( $x > 1/3$ ): view in (a)  $ab$   
 F9:2 plane and (b) three dimensions. AFM in the notation for the spin configuration is defined when two Fe atoms in one unit cell have  
 F9:3 antiparallel spins. Circles with solid and dashed outlines in (a) represent two Fe layers at  $c = 3/4$  and  $c = 1/4$ . Dark and light colors  
 F9:4 denote spins up and down. Solid rectangle depicts the smallest magnetic unit cell. (c) Plots of three equivalent domain directions within  
 F9:5 one Fe triangular lattice layer.

605 feature of magnetic peak at  $\mathbf{Q}_1 = (0.5, 0.5, 0)$  and the  
 606 second phase transition, one possible scenario is to assign  
 607 in-plane component associated with zigzag configuration

(Table II:  $\psi_5$  in  $\Gamma_2$ ), which is allowed by the group theory  
 and IR analysis. However, the calculated tilting angle  
 (see the Appendix) contradicts the large  $c$ -axis magnetic

608  
 609  
 610



F10:1 FIG. 10. Observed versus calculated intensities of nuclear (filled symbols) and magnetic (empty symbols) peaks at  $T = 38$  K (a) and  
 F10:2 5 K (b) for  $\text{Fe}_x\text{NbS}_2$  crystal with  $x = 0.34$ , (c)  $x = 0.32$ , (d)  $x = 0.33$ , and (e)  $x = 0.35$ . Symbols in different types are data collected in  
 F10:3 different scattering planes and instruments according to the legends in each panel. The calculated and observed intensities of peaks under  
 F10:4 different scattering geometries have been scaled simultaneously in order to be presented within the same frame.

611 anisotropy found in our susceptibility measurements and,  
 612 furthermore, would require a DM interaction orders of  
 613 magnitude larger than that allowed for by theory.

614 Alternatively, the rise-and-fall feature can be viewed as  
 615 simply the zigzag phase developing at the expense of the  
 616 stripe phase. This can readily occur with decreasing  
 617 temperature when the energy of two magnetic phases is  
 618 nearly degenerate and the relative energies of the two  
 619 phases change subtly as a function of temperature. That is,  
 620 the delicate energy balance between the two phases  
 621 changes around  $T_{N2}$  so that increasing regions of the  
 622 sample favor the zigzag phase as the temperature is  
 623 decreased. This can also happen if, as the zigzag phase  
 624 grows, the domain boundaries of the stripe phase are  
 625 converted to the zigzag configuration. Real space imaging  
 626 of the domains would help elucidate this growth process.  
 627 The redistribution of two magnetic phases is consistent  
 628 with the rounding of the  $T_N$  [Figs. 4(a) and 4(b)],  
 629 indicating a small spread in the Fe ratio across the sample.  
 630 In this scenario, the calculated intensities with ratio of  
 631  $\sim 75\%$  and  $\sim 35\%$  stripe phase for  $x = 0.33$  and  $x = 0.34$   
 632 samples, respectively, are consistent with the observed  
 633 patterns [Figs. 10(b) and 10(d)] at 5 K. The ordered  
 634 moment is extracted as  $3.2(3)\mu_B$  and  $3.5(3)\mu_B$  correspond-  
 635 ingly. The smaller value of these moments from the  
 636 saturated moment under high field ( $\sim 4\mu_B$  per Fe) is likely  
 637 due to errors in the normalization factor because of the  
 638 limited number of nuclear Bragg peaks.

639 In summary, the sample with measured  $x = 0.32$  shows a  
 640 pure stripe magnetic phase, the samples with  $x = 0.33$  and  
 641  $0.34$  show mixed phases, and the sample with  $x = 0.35$   
 642 shows a pure zigzag phase. This suggests that the crossover  
 643 from the stripe to the zigzag phase occurs at  $x = 1/3$ . The  
 644 subtle change of Fe ratio surprisingly results in a rapid  
 645 change of magnetic ground states, as well as the spintronic  
 646 response. Next, we discuss these findings and relations  
 647 between the two.

## 648 IV. DISCUSSIONS

### 649 A. Highly degenerate magnetic phases

650 In general, one finds that magnetic defects typically  
 651 suppress transition temperatures and reduce magnetic  
 652 correlations. Here, both the ordered moments and transition  
 653 temperatures are slightly reduced for off-stoichiometric  
 654 samples. The remarkable observation here is the dramatic  
 655 difference in the spin structures tuned by a small change in  
 656 the concentration and the nature of the magnetic defects,  
 657 namely from vacancies at the  $2c$  site to interstitials  
 658 (possibly at the  $2d$  or  $2b$  sites). As shown in the schematic  
 659 phase diagram (Fig. 11), our single crystal neutron dif-  
 660 fraction measurements reveal that the spin structure  
 661 changes from purely stripe to purely zigzag by varying  $\delta$   
 662 from  $\sim -0.01$  to  $0.02$  with both phases coexisting in near-  
 663 stoichiometric samples. We should emphasize that the two

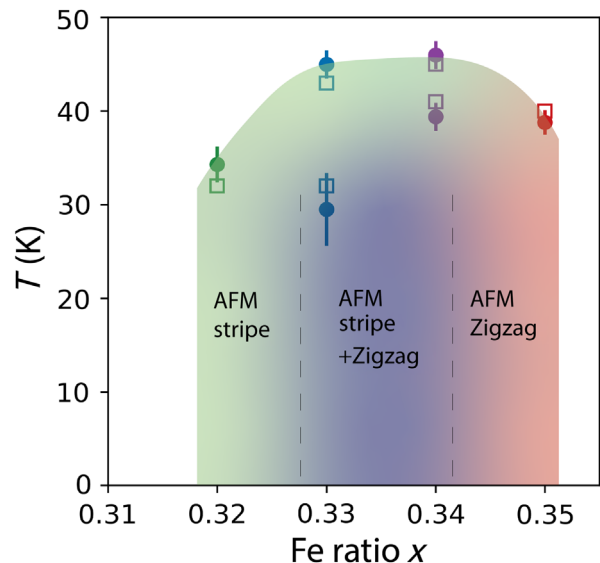


FIG. 11. Schematic phase diagram as a function of Fe ratio  $x$  around the critical value of  $1/3$ . This reveals a rapid change of magnetic phases from a single stripe order (green) in  $x < 1/3$ , across the coexisted two magnetic phases (purple) in  $x \sim 1/3$ , to a pure zigzag order (red) in  $x > 1/3$ . The solid dots denote the extracted transition temperatures from the neutron scattering measurements. The empty squares are characteristic temperatures of  $T_N$  (or  $T_f$ ) from the magnetization measurements guided to the eyes. The dashed lines are imaginary phase boundaries. For  $x = 0.31$ , we do not observe any detectable magnetic signal.

664 distinct spin structures reveal a totally different in-plane  
 665 spin configuration, which is uncommon in lamellar struc-  
 666 tures. These results demonstrate the first example of  
 667 flexible tuning of the magnetic ground state by a subtle  
 668 change of magnetic defects in the intercalation complexes  
 669 of the Nb and Ta dichalcogenides and, more generally, a  
 670 rare example in magnetic vdW compounds.

671 In the noncentrosymmetric intercalation species, several  
 672 characteristic magnetic interactions are relevant. Two  
 673 anisotropic exchange interactions are considered in  
 674  $\text{Fe}_{1/3+\delta}\text{NbS}_2$ . First, single-ion anisotropy and, possibly,  
 675 anisotropic exchange ( $\sim 2$  meV) result in highly anisotropic  
 676 uniaxial Ising behavior, distinct from the easy-plane  
 677 anisotropy observed in other  $\text{T}_x\text{NbS}_2$  species studied so far [14].  
 678 Second, the Dzyaloshinskii-Moriya antisymmetric  
 679 interaction originates from the loss of inversion symmetry.  
 680 Specifically, the interlayer DM interactions with an in-  
 681 plane component [59] could theoretically produce a small  
 682 in-plane moment. Unfortunately, the extreme sensitivity  
 683 of the magnetic ground state to the Fe concentration  
 684 makes the determination of any small tilt angle of the  
 685 spins indicated by zero-field anisotropic magnetoresistance  
 686 (ZFAMR) measurement [40] extremely difficult.

687 Both the Ruderman-Kittel-Kasuya-Yosida interaction  
 688 and the superexchange interaction were considered as  
 689 the relevant mechanisms for the magnetic ordering in this

690 system [13,14]. The former is long-ranged and variable in  
 691 both sign and magnitude; it relies on the separation of  
 692 localized moments and the Fermi wave vector [60–62]. The  
 693 latter is relatively short-ranged and the sign and magnitude  
 694 are often determined by application of the Goodenough-  
 695 Kanamori rules [63,64]. Since two dramatically different  
 696 ordered phases are facilitated by a small concentration of  
 697 magnetic defects, the superexchange interaction would be  
 698 barely affected without a change of the local structure.  
 699 Alternatively, the change of magnetic defects from vacan-  
 700 cies to interstitials, presumably, could influence the  
 701 RKKY interaction, especially for the interlayer exchange  
 702 coupling with its larger Fe-Fe distance. This scenario is  
 703 embedded in the oscillatory character of RKKY interaction  
 704 [60–62,65,66]; it is analogous to the alternating exchange  
 705 couplings in transition-metal layers separated by a non-  
 706 magnetic metal spacer [67,68].

707 Recent density functional theory studies [47,69] of the  
 708 AFM stripe and AFM zigzag magnetic ground states  
 709 strongly support our experimental results. To partially  
 710 account for enhanced localizations of Fe *d* electrons, a  
 711 Hubbard *U* correction was added in the DFT calculations.  
 712 Both  $U = 0.3$  eV and  $U = 0.9$  eV predict an easy-axis  
 713 anisotropy along [001], consistent with experiment.  
 714 PBE + *U* energies for magnetic orderings corresponding  
 715 to AFM stripe [ $\mathbf{k}_1 = (0.5, 0, 0)$ ] and AFM zigzag [ $\mathbf{k}_2 =$   
 716  $(0.25, 0.5, 0)$ ] are reported to differ in energy by at most a  
 717 few (1–3) meV per Fe atom. For context, this energy scale  
 718 is significantly smaller ( $2 \text{ meV}/k_B = 23.2 \text{ K}$ ) than the  
 719 onset temperature of either magnetic phase for near-  
 720 stoichiometry samples, rendering the stripe and zigzag  
 721 phases effectively degenerate. Moreover, intriguingly, the  
 722 relative energy ordering of AFM stripe and AFM zigzag  
 723 phases switches in going from PBE + *U* with  $U = 0.3$  eV  
 724 to  $U = 0.9$  eV [47]. The AFM stripe is lower in energy by  
 725 0.9 meV/Fe by using  $U = 0.3$  eV, whereas the AFM  
 726 zigzag is lower by 2.5 meV/Fe by using  $U = 0.9$  eV.

727 The near degeneracy and competition between AFM  
 728 stripe and zigzag phases near stoichiometry can be further  
 729 understood by a minimal Heisenberg model [47,69],  
 730 neglecting the single-ion anisotropy since this contribution  
 731 cancels when calculating differences in energy between  
 732 [001] oriented collinear magnetic orders. We highlight the  
 733 results of the prior work related to our experiments in what  
 734 follows. The mean-field energy with classical spin *S* can be  
 735 written as [47]

$$\begin{aligned}
 H = E_0 &+ \sum_{\langle ij \rangle} J_1 S^2 + \sum_{\langle\langle ij \rangle\rangle} J_2 S^2 + \sum_{\langle ij \rangle'} J_1' S^2 \\
 &+ \sum_{\langle\langle ij \rangle\rangle'} J_2' S^2 + \sum_{\langle\langle\langle ij \rangle\rangle\rangle'} J_3' S^2, \quad (4)
 \end{aligned}$$

736 where one, two, and three pairs of brackets denote  
 737 Heisenberg exchange constants between equivalent nearest,  
 738 next-nearest, and third-nearest neighbor interactions,

TABLE III. Heisenberg spin exchange constants, in meV/Fe atom, calculated with PBE + *U* for  $U = 0.3$  eV and  $U = 0.9$  eV. Positive values ( $J > 0$ ) are AFM coupling constants in our notation, and negative ( $J < 0$ ) couplings are FM. The prime refers to interplanar couplings.

	$J_1$	$J_1'$	$J_2$	$J_2'$	$J_3'$
$U = 0.3$ eV	+0.76	+0.49	-0.20	-0.006	-0.07
$U = 0.9$ eV	+0.57	+0.28	-0.16	-0.14	-0.09

740 respectively, and the prime refers to interlayer interactions  
 741 [Figs. 8(a) and 9(a)]. PBE + *U*-derived Heisenberg  
 742 exchange constants for both *U* values examined from  
 743 Ref. [47] are given in meV per Fe atom in Table III.  
 744 Based on the AFM nearest-neighbor interactions  $J_1$  and  $J_1'$   
 745 alone, the mean-field energies for AFM stripe and zigzag  
 746 are degenerate, and are primarily responsible for the  
 747 antiferromagnetism within and between the layers in both  
 748 structures. The degeneracy is broken by the relative small  
 749 values of the next-nearest neighbor interactions  $J_2$  and  $J_2'$   
 750 as well as third-nearest neighbor interlayer interaction  
 751  $J_3'$ . The energy difference between the two phases is  
 752  $E_{\text{stripe}} - E_{\text{zigzag}} = 4J_2' S^2 - 4J_2 S^2 - 8J_3' S^2$  [47]. The AFM  
 753 stripe phase is then favored when  $|J_2'| > |J_2| + 2|J_3'|$ ,  
 754 whereas the AFM zigzag phase is energetically favored  
 755 when  $|J_2'| < |J_2| + 2|J_3'|$ .

756 The relative change in magnitude and even signs of three  
 757 exchange constants can be attributed to the high degeneracy  
 758 of the two magnetic phases. As a possible microscopic  
 759 mechanism, we note that, on the one hand, the interlayer  
 760 exchange interactions originated via RKKY mechanism are  
 761 weak due to the long separation distance ( $\sim 9\text{--}10 \text{ \AA}$ ) and  
 762 further have an oscillatory nature. On the other hand,  
 763 magnetic Fe defects that reside within or between layers  
 764 can give rise to changes in the Fermi surface. Our  
 765 preliminary photoemission work reveals a rapid change  
 766 of the Fermi surface size from  $x < 1/3$  to  $x > 1/3$ . This  
 767 provides evidence that magnetic defects would affect the  
 768 couplings between localized moments via the conduction  
 769 electrons. Accordingly, the values or even sign of three  
 770 exchange constants would be quite sensitive to *x*, leading to  
 771 the tuning between the two AFM phases by magnetic  
 772 defects from  $x < 1/3$  to  $x > 1/3$ . As a result of the nearly  
 773 degenerate states, the delicate balance of the two magnetic  
 774 phases, which are spatially separated in the  $x = 1/3$   
 775 sample, can be changed causing one phase to win over  
 776 the other leading to the rise-and-fall feature in the order  
 777 parameter curve. The knob could be subtle changes in the  
 778 RKKY interactions with decreasing the temperature, or  
 779 magnetoelastic interactions that would turn on when  
 780 magnetic ordering sets in for the two different phases.  
 781 Further calculations could elucidate the possible mecha-  
 782 nisms. Also, highly degenerate states in the metallic  
 783 bilayer triangular lattice would require more theoretical



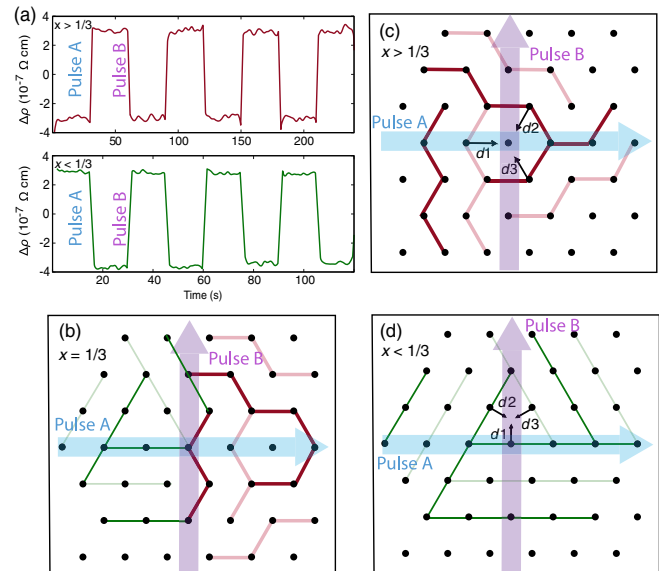
784 modeling beyond that for the frustrated magnetism in  
 785 the insulating single-layer triangular lattice Ising  
 786 antiferromagnet [70–72].

### 787 B. Relation to the spintronic features

788 Magnetic defects not only tune the magnetic ground  
 789 states, but they also influence the intriguing spintronic  
 790 features in  $\text{Fe}_{1/3+\delta}\text{NbS}_2$  [38,40]. By injecting a current  
 791 pulse along the [100] direction, for  $x < 1/3$  and  $x > 1/3$   
 792 samples, the change in transverse resistance is positive and  
 793 negative, respectively. In addition, the devices display more  
 794 active responses in off-stoichiometric samples, either below  
 795 or above the  $x = 1/3$  sample, while for  $x = 1/3$  the  
 796 amplitude of the resistance switching is dramatically  
 797 diminished. Our neutron work provides fundamental infor-  
 798 mation on the magnetic ground states in samples with  
 799 different Fe ratios that display rapid changes in the  
 800 spintronic behaviors as a function of Fe concentrations.

801 First, in the  $x < 1/3$  and  $x > 1/3$  sample our neutron  
 802 experiments clearly demonstrate single long-ranged stripe  
 803 order and zigzag order, respectively, with both revealing  
 804 three magnetic domains. These results provide important  
 805 indications for some of the observed switching features.  
 806 The observation of the change from stripe order to zigzag  
 807 order directly corresponds to the reversal of the sign of  
 808 the switching behaviors [Fig. 12(a), represented from  
 809 Ref. [40]]. Both stripe and zigzag phase have three  
 810 magnetic domains denoted as  $d_i$  ( $i = 1, 2, 3$ ). The  
 811 orientation of each magnetic domain  $d_i$  is defined as along  
 812 the direction of alternating spins. As shown in the con-  
 813 ceptual pictures in Figs. 12(c) and 12(d), if one assumes  
 814 that a current prefers a domain that is perpendicular to the  
 815 applied current pulse via the Rashba coupling [73], pulse A  
 816 (or B) will favor  $d_2 + d_3$  (or  $d_1$ ) domain in zigzag order  
 817 [Fig. 12(c)] and  $d_1$  (or  $d_2 + d_3$ ) domain in stripe order  
 818 [Fig. 12(d)]. This likely explains the opposite switching  
 819 responses in identical device geometries, or in other  
 820 words, under the same pulse current. Recent nonlocal  
 821 switching experiments [74] reveal a change of switching  
 822 behavior when populating another type of magnetic  
 823 domains in a secondary spot of the crystal, consistent  
 824 with this scenario.

825 Second, in the stoichiometric  $x = 1/3$  sample, the  
 826 evolution of the two magnetic phases is reflected in the  
 827 response to electrical current and magnetic field where both  
 828 the switching resistance and the ZFAMR reveal a sign  
 829 change upon lowering the temperature [40]. The sign  
 830 change upon cooling is consistent with the above specu-  
 831 lations, where the balance between two magnetic phases  
 832 will result in the preferences of different types of magnetic  
 833 domains induced by currents across the two transitions.  
 834 This is consistent with the calculation based on the current-  
 835 induced repopulation of magnetic domains in the  $x = 1/3$   
 836 sample [47]. In addition, the suppression of resistance



837 FIG. 12. (a) The change of resistance shows opposite signs  
 838 between  $x < 1/3$  and  $x > 1/3$  samples given two orthogonal  
 839 pulse currents A (blue) and (purple) (data from Ref. [40]). (b)–(d)  
 840 Conceptual pictures to illustrate the possible scenario for the sign  
 841 reversal of the switching resistance in the off-stoichiometric  
 842 samples as well as the suppressed magnitude in the  $x = 1/3$   
 843 sample, based on determined single stripe in  $x < 1/3$  and zigzag  
 844 in  $x > 1/3$  ordered phases, and coexistence of two phases in  
 845  $x = 1/3$ . For either stripe or zigzag phase, three magnetic  
 846 domains are plotted in green or red colors, respectively. Dark  
 847 and light colors denote spins up and down. Magnetic domain  
 848 orientations ( $d_i$  with  $i = 1, 2, 3$ ) are defined as the directions  
 849 along alternating spins. By the same pulse A (or B), two single  
 850 magnetic phases favor the populations of different domains. For  
 851 detailed descriptions, see text.

852 switching with decreasing temperature and compared to  
 853 the off-stoichiometric samples could then be simply  
 854 ascribed to the partial cancellation of the opposite  
 855 resistance changes where two phases coexist [Fig. 12(b)].  
 856 Our speculative ideas, however, require a detailed under-  
 857 standing of the actual switching mechanism both theo-  
 858 retically and experimentally, and this has not yet been  
 859 definitively identified.

860 Finally, the apparent absence of AFM order in our  
 861 heavily underintercalated sample ( $x = 0.31$ ) is surprising.  
 862 We do not observe any short-ranged magnetic peaks in our  
 863 measurements that are typically associated with spin-glass  
 864 order. Given the strong spin-glass-like feature in the  
 865 susceptibility measurements and spin transport behaviors,  
 866 it seems possible that dilute orphan spins might play a role  
 867 in enhancing the switching effects, but further studies of  
 868 samples in this regime by other experimental probes are  
 869 required to understand the absent AFM order. In all, we  
 870 provide empirical correlations between tunable magnetism  
 871 and spintronic features, providing the foundation for more  
 872 future work to decipher the mechanism.

858

## V. CONCLUSIONS

859 To conclude, we have performed single crystal neutron  
 860 diffraction experiments in the Fe intercalated transition-  
 861 metal dichalcogenide material  $\text{Fe}_{1/3+\delta}\text{NbS}_2$ , which recently  
 862 has been shown to exhibit intriguing resistance switching  
 863 and magnetic memory effects. Two long-range ordered  
 864 magnetic phases, specifically AFM stripe order with  
 865 wave vector  $k_1 = (0.5, 0, 0)$  and AFM zigzag order with  
 866  $k_2 = (0.25, 0.5, 0)$ , have been found and they can be  
 867 sensitively tuned by the Fe concentration as one goes from  
 868 the underintercalated to overintercalated region of the phase  
 869 diagram. This arises from the nearly degenerate energies  
 870 for the two spin structures, supported by our DFT calcula-  
 871 tion. Two phases can be tuned from one to the other due  
 872 to the oscillating nature of RKKY interaction  $J$  and the  
 873 competition between secondary intra- and interlayer inter-  
 874 actions. Two successive magnetic transitions are observed  
 875 in stoichiometric samples; the emergence of the second  
 876 magnetic phase is consistent with the remarkable near  
 877 degeneracy in energy of the two states. We provide crucial  
 878 information on magnetic ground states that form the basis  
 879 for understanding the interesting spintronic behaviors.  
 880 Our discovery of the highly tunable magnetic phases in  
 881 this bulk sample open up new, intriguing opportunities to  
 882 manipulate magnetic states and, concomitantly, the spin-  
 883 tronic properties by magnetic defects.

## ACKNOWLEDGMENTS

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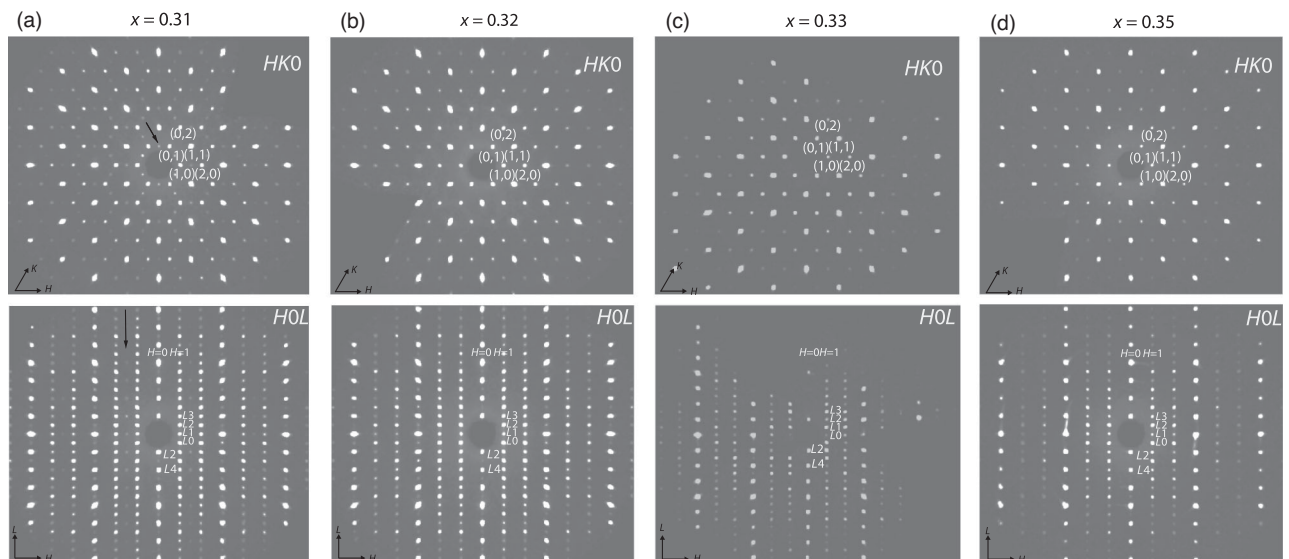
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## APPENDIX

This Appendix includes the following information: **3** (1) details on the single crystal x-ray diffraction and energy dispersive spectroscopy measurements, (2) characterization of the other single crystals that were used in the neutron diffraction experiments, (3) remarks on the possibility of a small in-plane moment developed below  $T_{N2}$  in the  $x = 1/3$  sample, and (4) the calculated pattern for each basis vector associated with  $k_1 = (0.5, 0, 0)$  and  $k_2 = (0.25, 0.5, 0)$ .

### 1. Details on the XRD and EDX measurements

The single crystal x-ray measurement was performed in small crystals from the same batch of  $x = 0.31, 0.32, 1/3$ , and  $0.35$  crystals as shown in Fig. 13. The images reveal three-dimensional Bragg peaks with no clear diffuse



F13:1 FIG. 13. Single crystal x-ray diffraction image in the  $HK0$  and  $HOL$  planes for  $x = 0.31$  (a),  $0.32$  (b),  $1/3$  (c), and  $0.35$  (d) crystals.  $L1$   
 F13:2 is short for  $L = 1$ ,  $L2$  for  $L = 2$ , etc. The black arrows denote the reflections associated with Fe occupying  $2b$  sites.

TABLE IV. The atomic coordinates, the Wyckoff positions, lattice parameters, and goodness of fits  $R_1$  from the single crystal structure refinements with the noncentrosymmetric space group  $P6_322$  for four different intercalation ratios  $x = 0.31, 0.32, 1/3,$  and  $0.35$  from top to the bottom table. Occ. stands for the occupation number. We used Olex2 for the structural analysis.

Atoms	$x$	$y$	$z$	Site	Occ.
Fe1	2/3	1/3	0.25	2c	0.75
Fe2	0	0	0.25	2b	0.1
Nb1	0	0	0	2a	1
Nb2	1/3	2/3	0.001 03(3)	4f	1
S	0.331 50(13)	0.001 67(15)	0.370 68(12)	12i	1
$a = b = 5.7608(1) \text{ \AA}, c = 12.1308(3) \text{ \AA}, R_1 = 2.9\%$					
Atoms	$x$	$y$	$z$	Site	Occ.
Fe1	2/3	1/3	0.25	2c	0.855
Fe2	0	0	0.25	2b	0.045
Nb1	0	0	0	2a	1
Nb2	1/3	2/3	0.001 15(3)	4f	1
S	0.330 96(14)	0.001 91(16)	0.370 82(12)	12i	1
$a = b = 5.7614(1) \text{ \AA}, c = 12.1436(3) \text{ \AA}, R_1 = 3.11\%$					
Atoms	$x$	$y$	$z$	Site	Occ.
Fe1	2/3	1/3	0.25	2c	0.9
Fe2	0	0	0.25	2b	0.023
Nb1	0	0	0	2a	1
Nb2	1/3	2/3	0.001 21(2)	4f	1
S	0.330 63(9)	0.002 14(11)	0.370 80(8)	12i	1
$a = b = 5.7596(3) \text{ \AA}, c = 12.1535(9) \text{ \AA}, R_1 = 3.44\%$					
Atoms	$x$	$y$	$z$	Site	Occ.
Fe1	2/3	1/3	0.25	2c	0.96
Fe2	0	0	0.25	2b	0.063
Nb1	0	0	0	2a	1
Nb2	1/3	2/3	0.001 24(2)	4f	1
S	0.330 66(17)	0.002 35(9)	0.371 21(6)	12i	1
$a = b = 5.7597(2) \text{ \AA}, c = 12.1914(4) \text{ \AA}, R_1 = 1.59\%$					

921 scattering signal along the  $L$  direction. A clear series of  
 922 ( $10L$ ) peaks imply a minimal disorder [30]. The structures  
 923 are well described by the noncentrosymmetric space group  
 924  $P6_322$  with refined structures listed in the Table IV by  
 925 using Olex2 structural analysis software. Additional weak  
 926 series of peaks (marked by black arrows) are associated  
 927 with the occupancy of Fe at  $2b$  Wyckoff sites. Equally  
 928 importantly, if alternative Fe sites occupy significantly, then  
 929 the crossover from the stripe to zigzag magnetic phase for  
 930 the  $2c$  Wyckoff sites would by necessity occur at an overall  
 931 Fe concentrations measurably higher than  $1/3$ .

932 Energy dispersive x-ray spectroscopy measurements were  
 933 performed to extract the iron intercalation ratio (Fig. 14). By  
 934 detecting  $\sim 20$  spots within the area of  $100 \times 100 \mu\text{m}^2$  for  
 935 each sample, we measured the concentration of Fe, Nb, and  
 936 S elements. We obtain the histogram for the value of  $x$  and

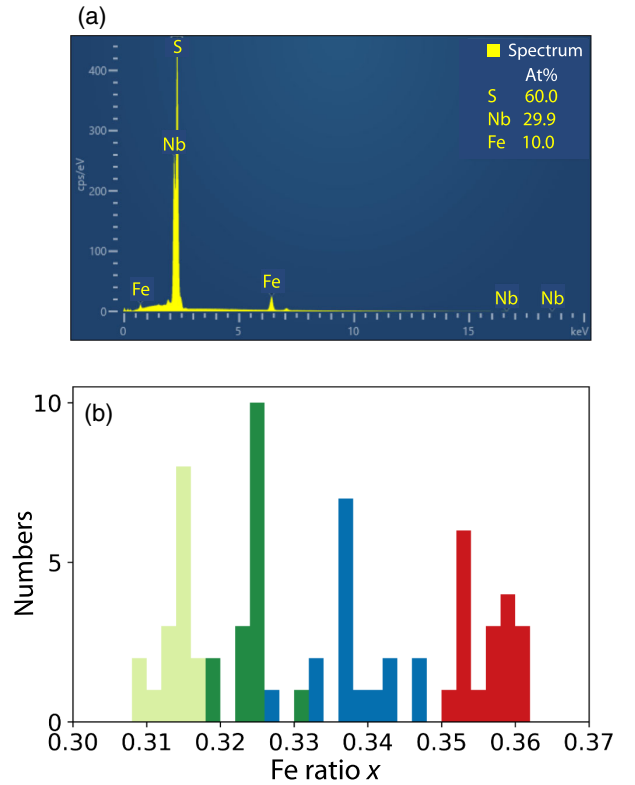
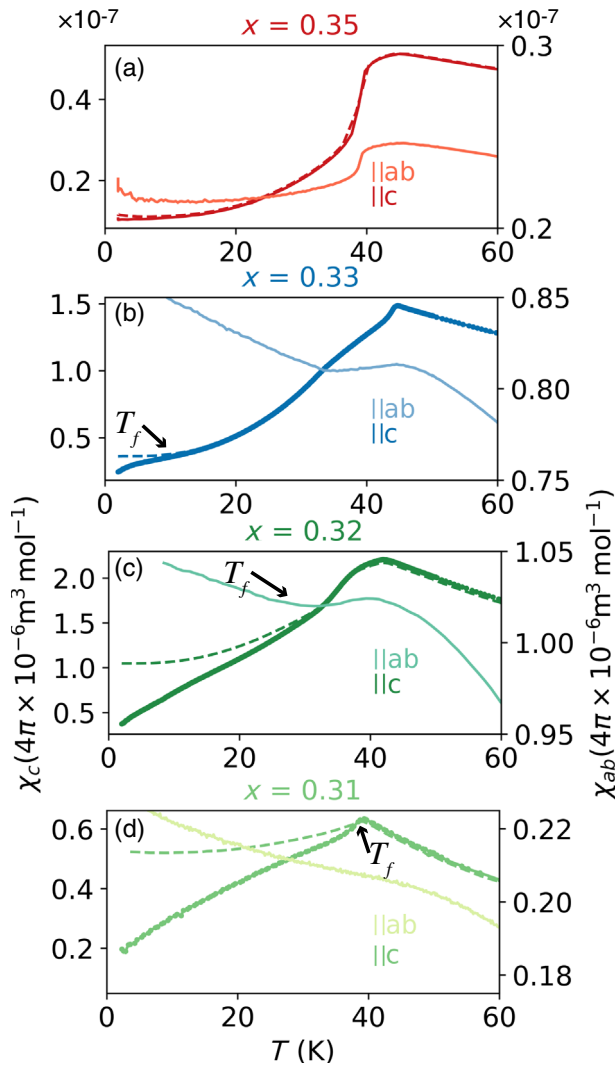


FIG. 14. (a) Representative energy dispersive spectroscopy  
 spectrum in  $x \sim 1/3$ . At% stands for atomic ratio. (b) The  
 histograms with binning size of 0.002 for Fe ratio  $x$  determined  
 from the EDX analysis from the  $x = 0.31$  (light green),  $x = 0.32$   
 (green),  $x \sim 1/3$  (blue), and  $x = 0.35$  (red) sample. The standard  
 deviation by measuring about 20 points is 0.003, 0.003, 0.005,  
 and 0.003 for  $x = 0.31, 0.32, 0.33,$  and  $0.35,$  respectively.

averaged intercalation ratio  $x$  for our measured crystals.  
 They are calculated to be  $x = 0.31, 0.32, 0.33, 0.34,$  and  
 $0.35$  given the EDX standard deviation of 0.003,0.003,  
 0.005,0.003. The estimated errors considering the factor  
 from the instrument is up to  $\pm 0.01$  for our crystals. The more  
 accurate ratio has been confirmed by inductively coupled  
 plasma optical emission spectroscopy measurements [38].

## 2. Magnetization measurements

The magnetization measurements for the other measured  
 neutron samples are shown in Fig. 15. The separation  
 between the zero-field-cooled and field-cooled susceptibil-  
 ity is dependent upon the deviation from the stoichiometric  
 ratio of  $1/3$ . ZFC and FC curves separate at a characteristic  
 temperature  $T_f$ ; such a separation does not occur in  
 samples with  $x$  very near  $1/3$ . In the crystal with  $x = 0.33$ ,  
 the susceptibility along the  $c$  axis exhibits one anomalous  
 peak followed by a broad hump with decreasing temper-  
 ature; correspondingly, two kinks in the in-plane suscep-  
 tibility  $\chi_{ab}$  are shown at  $T_{N1} \sim 32$  K and  $T_{N2} \sim 43$  K. In  
 the single crystal with  $x = 0.32$ , the second kink in  $\chi_{ab}$



F15:1 FIG. 15. Magnetization measurements for other compositions  
 F15:2  $x = 0.35$  (a),  $0.33$  (b),  $0.32$  (c), and  $0.31$  (d) with applied field of  
 F15:3  $0.1$  T along  $c$  axis and in  $ab$  plane. The dashed and solid lines  
 F15:4 corresponding to the measurements with field-cooled and zero-  
 F15:5 field-cooled process.

957 occurs around  $T_f \sim 32$  K. In the crystal with  $x = 0.31$ ,  $T_f$   
 958 is close to the peak anomaly in the  $c$ -axis susceptibility  
 959 around  $40$  K. In the  $x = 0.35$  sample, the characterization  
 960 data show the transition  $T_N \sim 40$  K. Both off-stoichiometric  
 961 and stoichiometric samples exhibit strong uniaxial  
 962 anisotropy in their susceptibilities (Fig. 2). For  $x > 1/3$ ,  
 963 Curie-Weiss fits to the susceptibility in the paramagnetic  
 964 region yield values of the paramagnetic effective moment  
 965  $\mu_{\text{eff}} = 5.0(3)\mu_B$  and Curie-Weiss temperature  $\theta_{\text{CW}} =$   
 966  $-50(2)$  K along the  $c$  axis;  $\mu_{\text{eff}} = 5.0(3)\mu_B$  and  $\theta_{\text{CW}} =$   
 967  $-165(5)$  K in the  $ab$  plane. For  $x < 1/3$ , the Curie-Weiss  
 968 fits in the paramagnetic region yield the values of para-  
 969 magnetic effective moment  $\mu_{\text{eff}} = 5.0(3)\mu_B$  and Curie-  
 970 Weiss temperature  $\theta_{\text{CW}} = -24(1)$  K along the  $c$  axis;  
 971  $\mu_{\text{eff}} = 4.7(2)\mu_B$  and  $\theta_{\text{CW}} = -104(2)$  K in the  $ab$  plane.

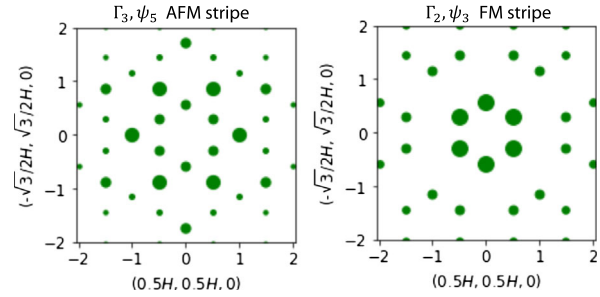


FIG. 16. Calculated intensities for given irreducible representa-  
 tion and basis vector associated with  $\mathbf{k}_1 = (0.5, 0, 0)$  and other  
 two equivalent  $k$ 's, describing a spin configuration of AFM stripe  
 (left) and FM stripe (right) with moment direction along  $c$  axis.  
 The size of dots represents the intensities of peaks, including  
 contributions of all equivalent domains.

F16:1  
 F16:2  
 F16:3  
 F16:4  
 F16:5  
 F16:6

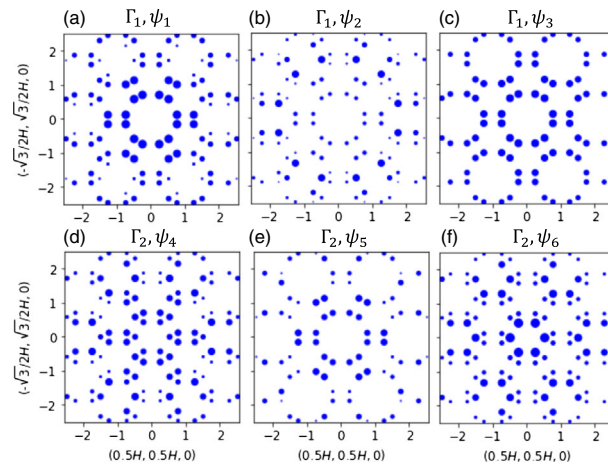


FIG. 17. Calculated intensities for given irreducible representa-  
 tion and basis vector associated with  $\mathbf{k}_2 = (0.5, 0.25, 0)$  and  
 other five equivalent  $k$ 's. Plots of (a)–(c) present the simulation  
 for  $\Gamma_1$  and (d)–(f) for  $\Gamma_2$  IR, corresponding to the magnetic space  
 group symmetry  $P_c2_12_12$  and  $P_c2_12_12_1$ , respectively. The size  
 of dots represents the intensities, including contributions of all  
 equivalent domains.

F17:1  
 F17:2  
 F17:3  
 F17:4  
 F17:5  
 F17:6  
 F17:7

### 3. Remarks on the magnetic structure analysis

In the samples with  $x \sim 1/3$  because of the complica-  
 tions presented by the coexistence of two different mag-  
 netic structures, it is not possible to say anything  
 meaningful about any in-plane moment. However, any  
 such in-plane moment would be caused by the interlayer  
 DM interaction which is small compared to both the  
 primary exchange and the  $c$ -axis anisotropy.

The single crystal neutron diffraction intensities are  
 calculated according to Eqs. (2) and (3) in the main text.  
 We utilize the BVs vector that describe the spin configu-  
 ration and calculate the magnetic intensities. We derived  
 the selection rule for each propagation wave vector as  
 $\delta_{2h}\delta_k$  for the wave vector  $\mathbf{k} = (0.5, 0, 0)$  domain,  
 and  $\delta_{2h}\delta_{h+2k}$  for the

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986 wave vector  $\mathbf{k} = (0.5, 0.25, 0)$  domain. Here  $h$  and  $k$  are  
 987 Miller indices for the wave vector transfer  $Q$ . The calculation  
 988 includes three domains with equal weights and the square of  
 989 the magnetic form factor. The normalization factor for the  
 990 magnetic peaks  $NC_m$  is obtained from the ratio between the  
 991 calculated square of the structure factor and the integrated  
 992 area of the nuclear peaks  $NC_n$ . The relation between them is  
 993  $NC_m = [(N_n V_m)/(N_m * V_n)] * NC_n$ , where  $V$  and  $N$  stand  
 994 for the volume and number of magnetic ( $m$ ) or nuclear ( $n$ )  
 995 unit cell, respectively. By this normalization, we can obtain  
 996 the ordered moment size by comparing the calculated and  
 997 measured intensities as shown in Fig. 10.

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